

1 **Highly active Fe-N-reduced graphene oxide electrocatalysts using sustainable amino acids**  
2 **as nitrogen source**

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10

11 **Abstract**

12 This work reports the preparation of oxygen reduction reaction catalysts by pyrolysis of a  
13 mixture of  $\text{Fe}(\text{NO}_3)_3$  and graphene oxide (GO) functionalized with three different amino acids,  
14 Arginine (Arg), Cysteine (Cys), Histidine (His). The combination of TPD, FTIR and XPS  
15 revealed that the functionalization of GO rendered a reduction of epoxide and carboxyl surface  
16 groups at along with the preferential generation of amide bridges. XPS confirmed the  
17 incorporation of nitrogen in edge and quaternary positions of the graphene layers after the  
18 pyrolysis at 800 °C of the amino acid functionalized GO mainly in form of pyridines (Arg),  
19 pyrroles (Cys) and quaternary nitrogen (His). XRD and XPS showed that most iron is retained  
20 after pyrolysis in the form of crystalline  $\text{Fe}_3\text{O}_4$  and in a lesser extend as  $\text{Fe}_2\text{O}_3$ . The amount,  
21 distribution, and particle size of iron species on the pyrolyzed samples relies on the amino acid  
22 used for the modification, with histidine rendering the lower retained amount and higher  
23 dispersion. Raman analyses pointed out that the structural order seems to follow the order  
24 His>Cys>Arg. The electrochemical characterization revealed that, in spite of having a lower iron  
25 content, Fe-rGO-His shows a higher double layer capacitance, improved conductivity and  
26 enhanced ORR activity than their counterparts, delivering onset and half-wave potentials only 30  
27 mV behind the current state-of-the-art platinum-carbon black catalyst. These results confirm that  
28 amino acids, and specially Histidine, can be interesting N-dopant agents for the preparation of  
29 nitrogen-iron graphene electrocatalysts with excellent ORR activities.

30

31 **Keywords:** Oxygen reduction reaction, Amino acids, Graphene, nitrogen doping, iron

32

## 33 **1. Introduction.**

34 Over the past decade, a strong research effort has been focused on the Oxygen reduction  
35 reaction (ORR). ORR is undeniably at the core of energy storage and conversion systems, such  
36 as metal-air batteries and fuel cells fuel. However, the industrial development of energy  
37 conversion systems has been hindered by the high cost of platinum, which is the metal of choice  
38 for the production ORR electro catalysts owing to its outstanding activity. In this sense, the  
39 scarcity and high costs, due to its limited reserves and the increasing demand in the market,  
40 along with several performance issues, such as the activity decline by low stability or under fuel  
41 crossover scenarios, greatly hampers the commercial success of energy devices based on ORR.  
42 To date many efforts have been made to overcome these problems, such as decreasing the  
43 platinum loading by the production of nanoparticles and/or alloys, or replacement of platinum by  
44 transition metals or even metal free carbon-based catalyst[1-4].

45 Today, Graphene (GR) has stood in the spotlight of research related to materials science,  
46 physics, chemistry, and nanotechnology. Ideally, GR is a two-dimensional carbon structure as  
47 thick as an atom that has a honeycomb-like structure[5]. However, several nanostructured carbon  
48 materials sharing a similar structure but with different degrees of structural order and surface  
49 chemistry composition are also regarded as graphene, like exfoliated graphite by mechanical  
50 methods, few-layered graphene by chemical vapor deposition or reduced graphene oxide [6]. GR  
51 properties relies on the preparation method. For instance, exfoliated graphite shows  
52 extraordinary properties, such as excellent thermal and electrical conductivity, high optical  
53 transparency, high surface area and Young modulus [7-10], but massive GR production is not  
54 achievable following this procedure.

55 In this sense, the chemical-thermal reduction of graphene oxide (GO) can be highlighted as  
56 one of the most viable production pathways. Graphene oxide has a hexagonal structure  
57 resembling the one of GR, but it contains a large number of surface oxygen groups, such as  
58 epoxides, hydroxyls and carboxylic acid moieties [11]. The most common method of producing  
59 GO is the Hummers chemical method [12, 13], consisting of the oxidation of graphite using  
60 protonated solvents, which delivers the formation of graphite oxide. The basic unit structure of  
61 Graphite oxide is GO, and therefore GO can be obtained by delamination of graphite oxide using  
62 mechanical methods. The presence of oxygen groups in GO has benefits and drawbacks. They  
63 increase the solubility and processability of GO compared to GR and enables a wide number of  
64 functionalization routes. However, they also neglect some of the unique GR properties, such as  
65 electrical conductivity. The functional groups of GO are removed by reduction, generating  
66 reduced graphene oxide (rGO), which is a family member of GR materials [14]. There are a large  
67 number of methods available for reducing GO to produce Gr or rGO, from where thermal or  
68 chemical reduction are the most widely used due to their ease.

69 Among these methods, the treatment with amino acids (AAs) have been used not only to  
70 reduce GO but also to functionalize Gr and GO, and even to produce Gr through graphite  
71 exfoliation by AA intercalation[15]. In this regard, Chwalibog *et al.* achieved significant and  
72 positive results in their research work on Gr functionalized with arginine for the treatment of  
73 glioblastoma multiform [16]. In another research work, Shabani *et al.* studied the reduction and  
74 functionalization of GO by the L-histidine amino acid [17]. Moreover, the production of N-  
75 doped carbon materials as sustainable ORR electro catalysts via AA functionalization of GO  
76 have been proposed using cysteine and arginine [18-21].

77        Interestingly, the activity of N-doped carbon materials for ORR increases when combined  
78        with transition metals, showing high activity, high selectivity to water and strong resistance to  
79        poisoning and fuel crossover [22]. Between them, the Fe-N pair have been proposed as an  
80        outstanding active site for ORR, being the most promising alternative to platinum in terms of  
81        availability, cost and activity [23]. Synthesis of Fe-N catalysts usually follow two different  
82        pathways, namely i) loading of iron nanoparticles or iron precursor on top of N-doped carbon  
83        materials followed by thermal reduction [24] and ii) pyrolysis of iron and nitrogen-rich  
84        precursors on the surface of carbon materials [25]. To the date, there are only a few works  
85        dealing with the preparation of Fe-N ORR catalysts using amino acids [26, 27]. However, to our  
86        knowledge, there is no report concerning the use of different amino acids for GO  
87        functionalization and subsequent production of Fe-N-rGO catalysts by thermal treatment in the  
88        presence of iron precursor. This synthesis could be a sustainable approach to produce  
89        nanostructured rGO catalysts for the oxygen reduction reaction [27-31].

90        In this work, iron loaded N-doped reduced graphene oxide were prepared by functionalization  
91        of GO using three different amino acids, cysteine, histidine, and arginine followed by pyrolysis  
92        in the presence of iron nitrate. The porosity, surface chemistry and structure of the resulting  
93        materials have been characterized, and their activity towards ORR in acid and alkaline  
94        electrolyte has been evaluated.

95

## 96 2. Materials and Methods

### 97 2.1. Synthesis of electrocatalysts

98 GO was first synthesized through Hummers method [10]. Briefly, natural graphite flakes  
99 (Asbury Graphite Mills, US) and sulfuric acid ( $\text{H}_2\text{SO}_4$ , 95.5–96.5%) were mixed and stirred at  
100 200 rpm. Fuming nitric acid ( $\text{HNO}_3$ , 65%) was added into the mixture and stirred for 24 h at  
101 room temperature. Then, deionized water was added slowly, and the obtained mixture was  
102 washed and dried at 60 °C to obtain graphite intercalation compounds (GIC). To obtain  
103 Expanded Graphite (EG), the dry GIC powder was thermally expanded at 1050° C for 15 s. EG  
104 maintains the graphite layered structure, but the interlayer space can expand to more than 150  
105 times. Therefore, oxidizing agents for the preparation of GO will diffuse faster between the  
106 layers of expanded graphite. Then, 1 g of EG and 200 ml of sulfuric acid were mixed and stirred.  
107 After that, 10 g of  $\text{KMnO}_4$  was added dropwise. The obtained mixture was transferred into an ice  
108 bath and 200 ml of deionized water (DW) and 50 ml of  $\text{H}_2\text{O}_2$  was added, the color of the mixture  
109 changed to light brown. The GO particles were stirred, washed with HCl later centrifuged.

110 Arginine, cysteine and histidine were selected for GO functionalization because of their high  
111 content of N in arginine (Arg), the presence of both N and S in cysteine (Cys), and the presence  
112 of imidazole group and N in histidine structure (His). A GO:AA ratio of 1:2 was selected for  
113 GO:AA [32-34]. The details of the synthesis of catalysts are as follow: first, 80 mL of GO  
114 solution containing 100 mg of dissolved GO was placed on a stirrer. 200 mg of the amino acids  
115 were dissolved in 10 mL of DW separately in a beaker and later added to the GO solution. The  
116 GO-AA solution was placed on a stirrer for 8 hours at room temperature (25 °C). For recovering  
117 the solids, the solutions were centrifuged at 18000 rpm for 10 minutes (This action was repeated  
118 three times), filtered using a nylon membrane (pore size 430 nm, Millipore) and dried at 80 °C.

119 The GO-AA solids were recovered at this point in the form of brittle flakes and stored for further  
120 characterization. These samples have been named as GO-Arg, GO-Cys and GO-His in  
121 accordance with the amino acid employed during the functionalization.

122 For the preparation of the Fe-N-rGO catalysts, GO-AA samples served as support, whereas Fe  
123  $(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  was chosen as the iron source. A solution of Fe  $(\text{NO}_3)_3$  (0.1 M) in distilled water  
124 was added to GO-AA solutions containing the different amino acids (GO:AA:  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$   
125 ratio of 1:2:0.66) and mixed on a stirrer at room temperature for 24 hours. The resulting solution  
126 was then centrifuged at 18000 rpm for 10 minutes to remove the excess of water. The recovered  
127 wet solid was then placed in an alumina crucible, dried at 50 °C overnight and heated up inside a  
128 furnace at 800 °C (heating rate: 10 °C/min, holding time: 10 minutes) under  $\text{N}_2$  atmosphere (flow  
129 rate: 150 mL/min STP) to pyrolyze the mixture and obtain Fe-N-C catalysts [35]. The final solids  
130 have been named as Fe-rGO-Arg, Fe-rGO-Cys and Fe-rGO-His, depending on the amino acid  
131 used in the functionalization step. For comparison purposes, iron-free GO-AA samples were also  
132 pyrolyzed at 800 °C under the previously described conditions to obtained reduced graphene  
133 oxide samples, labelled as rGO-AA.

### 134 **2.3. Characterization**

135 In this study, the elemental composition of the samples was analyzed in a TruSpec micro  
136 CHNSO (Leco) to determine the mass fractions of carbon, hydrogen, nitrogen and sulfur. The  
137 surface chemistry of the samples was analyzed through X-ray photoelectron spectroscopy (PHI  
138 5000 VersaProbe II model, Physical Electronics) with monochromatic Al k-alpha radiation,  
139 beam being focused in a 10-micrometer window. The whole spectrum was shifted using as  
140 reference the position of the C1s peak, which was set to 284.5 eV. Fourier-transform infrared  
141 spectroscopy (FTIR, SHIMADZU, 8400S model) was used to check the formation of functional

142 groups in the structure of GO and the doping of the desired metal. The amount and type of  
143 oxygen surface groups were analyzed by Temperature Programmed Desorption (TPD), which  
144 was performed in a custom quartz reactor by heating 15-30 mg of GO and GO-AA samples from  
145 room temperature up to 900 °C under N<sub>2</sub> flow (150 mL/min STP) using a heating rate of 10  
146 °C/min. The desorption rate of CO and CO<sub>2</sub> during this thermal run was measured using NDIR  
147 detectors (Ultramat 23, Siemens). Furthermore, X-ray diffraction patterns (XRD) of the samples  
148 were recorded using a 2θ range between 5 and 90° in an EMPYREAN diffractometer  
149 (PANalytical), using CuKα monochromatic radiation operating at 45 kV and 40 mA. Raman  
150 spectra of the samples were registered in an Invia Qontor Raman Confocal microscope  
151 (RENISHAW), using the 532 nm laser line (grating: 240 L/mm, slit opening of 50 μm) and a  
152 spectral resolution of ±1 cm<sup>-1</sup>. The power of the beam was set to 0.27 mW to avoid any  
153 modification of the samples during the measurement. The morphology of the samples was  
154 studied by transmission electron microscopy (TEM) images with a JEOL JEM-2010 microscope  
155 operated at 200 kV. In addition to the above diagnostic analyses, electrochemical  
156 characterization and activity tests related to ORR were performed.

#### 157 **2.4. Electrochemical test**

158 The electrochemical characterization of the samples was carried out using an SP-200 potentiostat  
159 (biologic) in 3-electrode cell configuration, where a graphite rod and saturated calomel electrode  
160 (SCE) were set as counter and reference electrode, respectively. The working electrode consisted  
161 in a rotating disk electrode (RDE, OrigaTrod, Origalis) made of glassy carbon (GC, 5 mm  
162 diameter). For the preparation of the working electrode assembly, 1 mg/mL ink is prepared by  
163 mixing 5 mg of the sample with 5 mL of a solution consisting in 5:1:0.5  
164 water:isopropanol:Nafion© (5% solution, Sigma Aldrich). The mixture is sonicated in ultrasound

165 bath for 20 minutes. The GC electrode was then decorated with the sample by pipetting 20  $\mu\text{L}$  of  
166 the ink on top of the GC surface (electrode loading of ca. 100  $\mu\text{g}/\text{cm}$ ) and drying the GC at 110  
167  $^{\circ}\text{C}$ .

168 ORR electrochemical activity tests were carried out at 25  $^{\circ}\text{C}$  in alkali electrolyte (0.1 M KOH).  
169 The electrochemical behavior of the working electrode was analyzed by performing 20 cyclic  
170 voltammetry (CV) cycles at 50 mV/s in the 0 to 1.2 V vs reversible hydrogen electrode (RHE)  
171 range using  $\text{N}_2$ -saturated electrolyte. CV and linear sweep voltammetry (LSV) from 0 to 1 V  
172 were performed at 5 mV/s in  $\text{N}_2$  and  $\text{O}_2$ -saturated electrolyte and under different rotating rates  
173 (from 400 to 2025 rpm). The number of electrons transferred during ORR,  $n$ , were determined  
174 using the Koutecky Levich (KL) equation, for which the following equation was used:

$$175 \quad \frac{1}{j} = \frac{1}{j_K} + \frac{1}{j_L} = \frac{1}{j_K} + \frac{1}{B \cdot n} \cdot \omega^{-1/2}$$

176 where  $j_K$  and  $j_L$  are the kinetic and mass transfer limited current densities,  $j$  is the experimentally  
177 observed current density,  $\omega$  is the angular velocity and  $B$  is the proportionality coefficient, which  
178 can be calculated as:

$$179 \quad B = 0.62 \cdot D_{\text{O}_2}^{2/3} \cdot C_{\text{O}_2} \cdot \nu^{-1/6} \cdot F$$

180 In this equation,  $D_{\text{O}_2}$  is the diffusion coefficient of oxygen,  $C_{\text{O}_2}$  is the bulk concentration of  $\text{O}_2$ ,  $\nu$   
181 is the kinematic viscosity of the electrolyte and  $F$  is the Faraday constant.

182 The electrochemical response and ORR activity of commercial 20% platinum on Vulcan XC72  
183 catalyst (Sigma-Aldrich) was also analyzed using the same protocol for comparison purposes.

184

185 **3. Results and discussion**

186 **3.1. GO functionalization with amino acids**

187 The composition of GO before and after functionalization with AA was determined by ultimate  
188 analyses, Table 1. GO was dried at 60 °C under vacuum prior for this measurement. The analyses  
189 confirm the presence of nitrogen in all GO-AA samples due to the successful linking of the  
190 biomolecules on the surface of graphene oxide. The loss of oxygen is an interesting side result  
191 that could be related to the chemical reduction of GO amino acids [15]. The presence of sulfur in  
192 GO-Cys is a desired feature of this sample and comes from the presence of thiol group in  
193 cysteine. Considering the amounts of nitrogen in the parent AAs and the absence of nitrogen in  
194 GO, the maximum theoretical nitrogen loading that can be achieved at 1:2 GO:AA for each  
195 sample would be 32.1, 11.6 and 27.1 N wt.% for GO-Arg, GO-Cys and GO-His, respectively.  
196 Considering the nitrogen amounts detected by ultimate analyses (Table 1), the estimated N  
197 fixation ratios were 38, 30 and 34 % for GO-Arg, GO-Cys and GO-His, respectively. The  
198 slightly higher incorporation of N by arginine fixation is probably due to the presence of  
199 terminal amino functions, which could form amide bridges with carboxylic moieties of GO [36].  
200 It is also important to note that the cysteine molecule has a N:S atomic ratio of 1:1, whereas the  
201 elemental analysis of GO-Cys sample showed an experimental ratio of 2:1, which could be  
202 related to the loss of the thiol group due to the partial decomposition of the AA after  
203 functionalization.

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205

206

**Table 1. Elemental composition of samples (% dry & ash free basis)**

Sample	C		H		N		S		O*	
	<i>wt</i> <sup>#</sup>	<i>at</i> <sup>\$</sup>	<i>wt</i>	<i>at</i>	<i>wt</i>	<i>at</i>	<i>wt</i>	<i>at</i>	<i>wt</i>	<i>at</i>
GO	42.4	38.8	2.1	23.1	0.0	0.0	0.0	0.0	55.5	38.1
GO-ARG	54.2	38.6	4.5	38.4	12.3	7.5	0.1	0.0	28.9	15.4
GO-CYS	55.7	42.7	3.8	34.9	3.5	2.3	4.0	1.1	33.0	19.0
GO-HIS	51.4	41.2	3.2	30.8	9.1	6.3	0.1	0.0	36.1	21.7
rGO-ARG	79.9	69.7	1.7	17.8	4.9	3.7	0.0	0.0	13.5	8.8
rGO-CYS	82.7	76.0	1.2	13.2	2.6	2.0	1.8	0.6	11.7	8.1
rGO-HIS	83.2	74.3	1.4	15.0	3.8	2.9	0.0	0.0	11.6	7.8
Fe-rGO-ARG	89.2	78.7	1.4	14.8	2.4	1.8	0.0	0.0	7.0	4.6
Fe-rGO-CYS	90.7	81.0	1.3	13.9	2.0	1.5	1.4	0.5	4.6	3.1
Fe-rGO-HIS	90.6	79.8	1.4	14.8	1.3	1.0	0.0	0.0	6.7	4.4

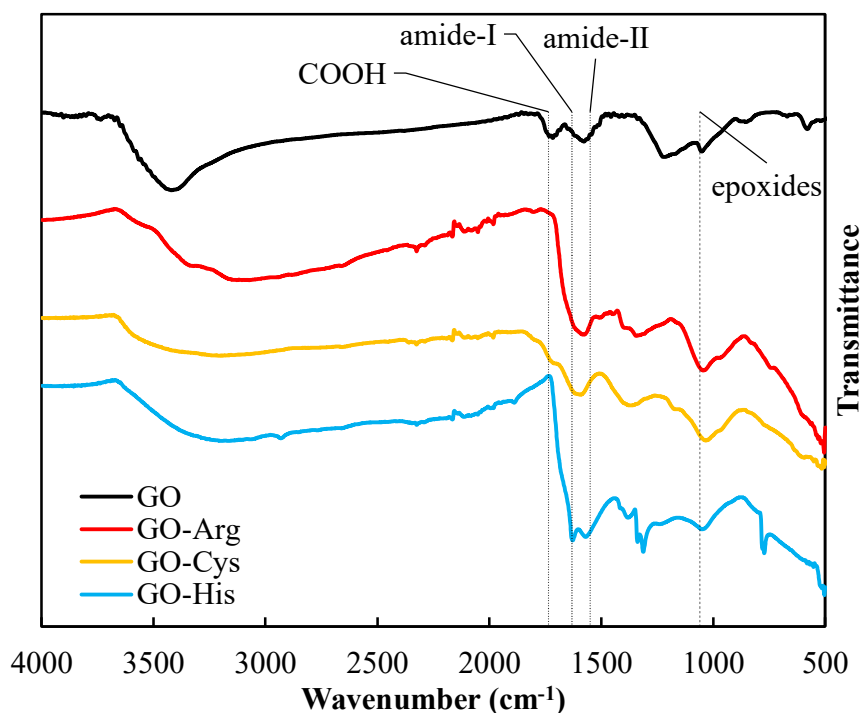
\*O= 100 -C -H -N -S – ash

<sup>#</sup>*wt*= weight concentration

<sup>\$</sup>*at*= atomic concentration

207 FTIR spectra of GO and GO-AA samples were recorded and are compiled in Fig. 1. It can be  
208 seen that GO shows as its main features the absorption bands related to the presence of carboxyl  
209 groups at 1735 cm<sup>-1</sup>, as well as the C-O stretching vibration from epoxy groups at 1060 cm<sup>-1</sup> and  
210 the broad band at 3250 cm<sup>-1</sup> owing to the presence of hydroxyl functionalities. After  
211 functionalization with amino acids, (GO-His, GO-Cys and GO-Arg in Figure 1), new bands  
212 emerge in GO-AA samples related to the presence of the respective amino acids, such as the

213 stretching vibration of C-N at  $1610\text{ cm}^{-1}$  and  $1335\text{ cm}^{-1}$  for histidine. Less significant peaks  
214 related to the different bonds of each amino acid are also observed.



215

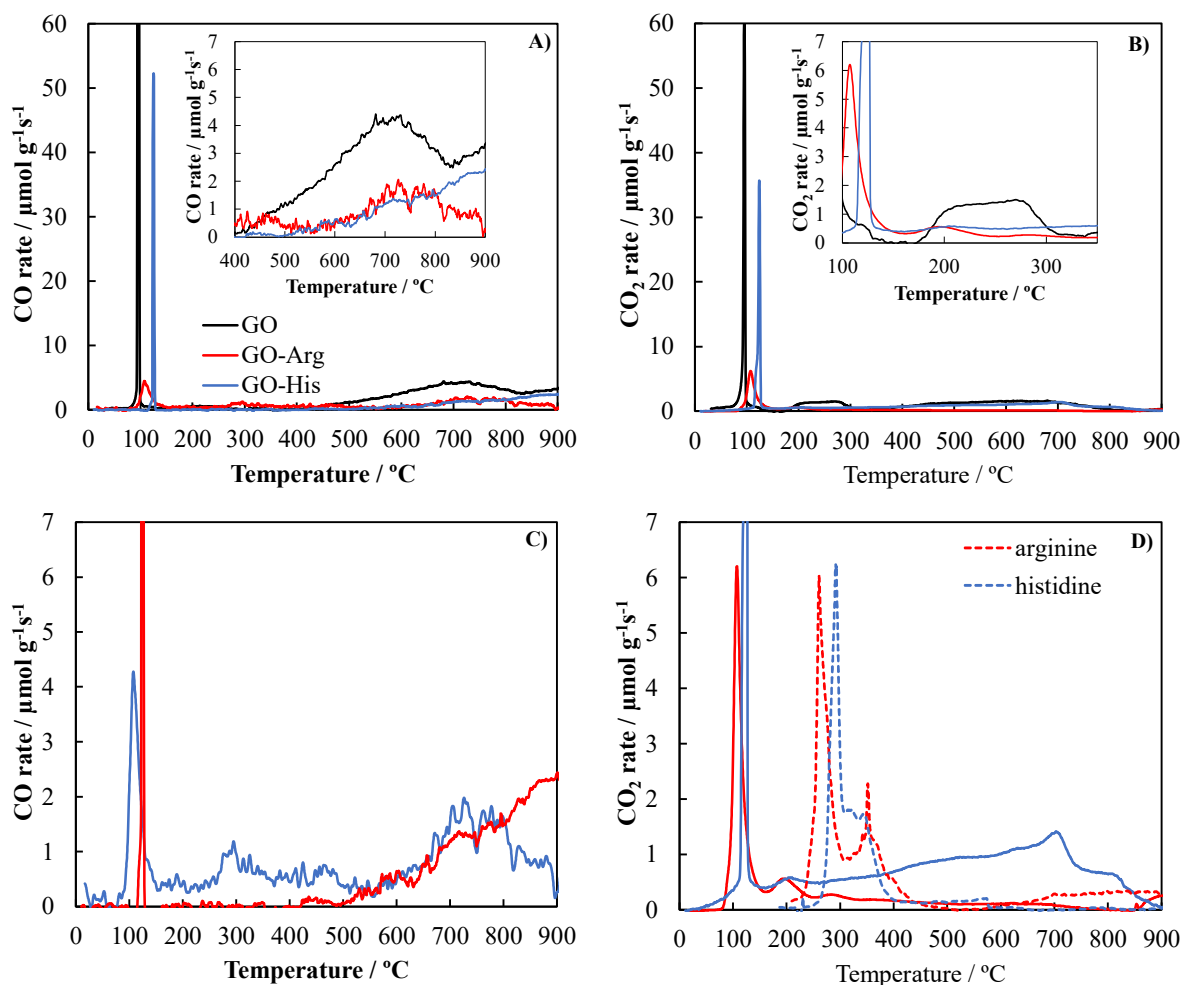
216 **Figure 1.** FT-IR spectra of GO, GO-Arg, GO-His and GO-Cys samples.

217 Interestingly, these spectra also confirm the formation of amide bridges through the development  
218 of amide-I and amide-II bands at around  $1630$  and  $1550\text{ cm}^{-1}$ , respectively [37], being especially  
219 clear for GO-His. In line with this, a strong decrease in the bands related to the carboxyl type  
220 functional groups is observed at  $1735\text{ cm}^{-1}$ , suggesting that the main functionalization pathway  
221 of GO with the involved amino acids proceeds through the formation of amide bridges via the  
222 reaction of the amine groups with GO carboxylic acid groups. The FTIR spectra of all the  
223 samples also reveals that the peak related to epoxy groups is severely modified, showing the  
224 contribution of C-O vibrations from the functional groups of the amino acids. In this line, others  
225 authors have also reported the reduction of GO using amino acids [38]. Specifically, Tran et al.

226 observed the loss of hydroxyl, carbonyl and epoxy groups during the deoxygenation of GO using  
227 aspartic acid and valine [39].

228 The CO and CO<sub>2</sub> temperature programmed desorption (TPD) profiles of GO, GO-cys, GO-Arg  
229 and GO-His were measured to study the thermal reduction of GO before and after the amino acid  
230 functionalization of these samples. For the sake of comparison, thermal decomposition of pure  
231 cysteine, arginine and histidine was also analyzed. TPD is a technique that allows to identify the  
232 surface oxygen groups in carbon materials attending to their different thermal stability and the  
233 formation of either CO or CO<sub>2</sub> [40].

234 Figure 2.a and 2.b compares the CO and CO<sub>2</sub> profiles, respectively, of GO before and after the  
235 functionalization with histidine and arginine. Cysteine functionalization showed similar profiles  
236 than GO-His and was not included for the sake of clarity. The TPD profiles of pristine GO shows  
237 high evolution of CO and CO<sub>2</sub> at ca. 100 °C in a narrow interval of temperatures, which is related  
238 to the presence of large quantities of labile and rather unstable epoxy and hydroxyl groups in the  
239 basal plane [41, 42]. The CO<sub>2</sub> evolution in the range of 200-300 °C is ascribed to the presence of  
240 carboxyl groups at edge sites, as reported for conventional carbon materials [40]. Between 400  
241 and 600 °C, the CO and CO<sub>2</sub> evolution rates of GO grow in parallel due to the thermal  
242 decomposition of anhydrides. Beyond temperatures of 600 °C, CO release still increases due to  
243 the decomposition of phenol and carbonyl groups at edge positions [40].



244

245 **Figure 2.** A) CO- and B) CO<sub>2</sub>-Temperature Programmed Desorption profiles of GO, GO-Arg  
 246 and GO-His. C) detailed CO-TPD and D) CO<sub>2</sub>-TPD profile of GO-Arg and GO-His, and the  
 247 parent amino acids (dotted lines)

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**Table 2.** CO and CO<sub>2</sub> evolution and oxygen content  
determined from TPD analyses

	CO	CO <sub>2</sub>	Total O	Total O
Sample	mMol / g	mMol / g	mMol / g	wt. %
GO	12.35	5.86	21.45	38.5
GO-Cys	3.65	3.25	10.15	16.2
GO-Arg	3.50	1.48	6.45	10.3
GO-His	3.78	4.39	12.56	20.1

252

253 Table 2 compiles the amount of CO and CO<sub>2</sub> evolved from GO before and after functionalization  
 254 with amino acids. The total amount of oxygen measured from this technique is lower than the  
 255 values obtained from ultimate analyses due to the presence of interstitial water in GO (which is  
 256 known to be the main released product at temperatures around 100 – 150 °C [41, 43]), water  
 257 coordinated with carboxylic functionalities and probably surface oxygen groups of large thermal  
 258 stability that remains on the carbon surface even at temperatures higher than 900 °C, such as  
 259 pyrone groups [44].

260 After functionalization, a strong decrease in the low temperature CO and CO<sub>2</sub> evolution peaks,  
 261 along with a temperature shift to higher values is observed. In addition, the carboxylic moieties  
 262 of GO at temperatures between 200 and 300 °C are lost, inset of Fig. 2.b, which seems to confirm  
 263 the FTIR results regarding preferential functionalization of amino acids via formation of amide  
 264 bridges. It is also important to note that AA functionalization produce a decrease in the CO  
 265 evolution in the range of 500 to 800 °C, inset of Fig. 2.a, which is likely related to side reactions  
 266 between the amino acids and hydroxyls. These side reactions between surface oxygen and

267 nitrogen groups might produce the formation of pyridine, pyrrole, and other nitrogen  
268 functionalities [36, 45], explaining the high nitrogen content observed for pyrolyzed samples in  
269 Table 1, and in accordance to XPS analyses (discussed below).

270 The chemical reduction of surface oxygen groups of GO by the amino acids is clearly observed  
271 in the decrease of CO, CO<sub>2</sub> and total O evolution after the functionalization of GO, shown in  
272 Table 2. However, functionalization with arginine delivers a stronger chemical reduction, as  
273 pointed out by the larger decrease on CO<sub>2</sub> evolution values observed in Table 2. Figure 2c and 2d  
274 shows the detailed TPD profiles (for CO and CO<sub>2</sub>, respectively) of GO-His and GO-Arg for a  
275 clearer understanding of their differences. It can be seen that stronger chemical reduction of GO  
276 caused by arginine is mostly related to the removal of anhydrides and lactones, since CO<sub>2</sub>  
277 evolution at temperatures higher than 300 °C is mostly suppressed, Figure 2.d. The slightly larger  
278 CO evolution at 250-500 °C for this sample, Figure 2.c, could be related to the decomposition of  
279 the carbonyl group of the amide bridges, as previously reported in the literature [36]. Finally,  
280 since the amino acids also have carboxylic functions in their structure, the CO<sub>2</sub> TPD profile of  
281 arginine and histidine are also included in Figure 2.d for comparison purposes (dotted lines). It is  
282 seen that the carboxylic acid groups of the amino acids decompose producing the release of  
283 carbon dioxide at temperatures between 260 and 400 °C. These desorption peaks are not  
284 observed in the GO-His or GO-Arg profiles, which can be due to changes in the stability and  
285 therefore in the thermal decomposition reactions of amino acids once they are linked to the  
286 surface of GO.

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### 289 **3.2. Characterization of Fe-rGO-AA electrocatalysts**

290 The chemical composition of Fe-rGO-AA electrocatalysts was determined by ultimate analyses,  
291 the results are compiled in Table 1. To evaluate the effect of the addition of iron in Fe-rGO-AA  
292 catalysts, GO-AA samples were also pyrolyzed at 800 °C to obtain N-doped ORR catalysts  
293 (rGO-AA in Table 1). The pyrolysis of GO-AA decreases the oxygen and nitrogen contents.  
294 However, the final content of nitrogen is in the range of 2.5-5 %wt, confirming the formation of  
295 nitrogen-rich carbon materials. These amounts are lower than those reported by Wang et al. in  
296 the production of N-doped carbon nanosheets, where histidine and cysteine are pyrolyzed in the  
297 presence of zinc nitrate, achieving N contents of 8.2 and 7.6 wt. %, respectively [46].  
298 Interestingly, sulfur doping is also attained in rGO-Cys sample, although the incorporation of  
299 nitrogen seems to be promoted over that of sulfur (N:S atomic ratio increases from 2.0 up to 3.3  
300 after pyrolysis). A similar finding was obtained by Zhang et al. when pyrolyzing polydopamine-  
301 cysteine functionalized rGO, attaining preferential N-doping on the resulting sample [47, 48].  
302 When pyrolysis is performed in the presence of iron species (Fe-rGO-AA samples in Table 1), a  
303 clear increase in carbon content is achieved, reaching values close to 90% for all the samples.  
304 This is an expected outcome, given that iron is a well-known graphitization catalyst [49].  
305 Unfortunately, the presence of iron nitrate suppresses partially the high nitrogen doped levels  
306 observed in iron-free samples.

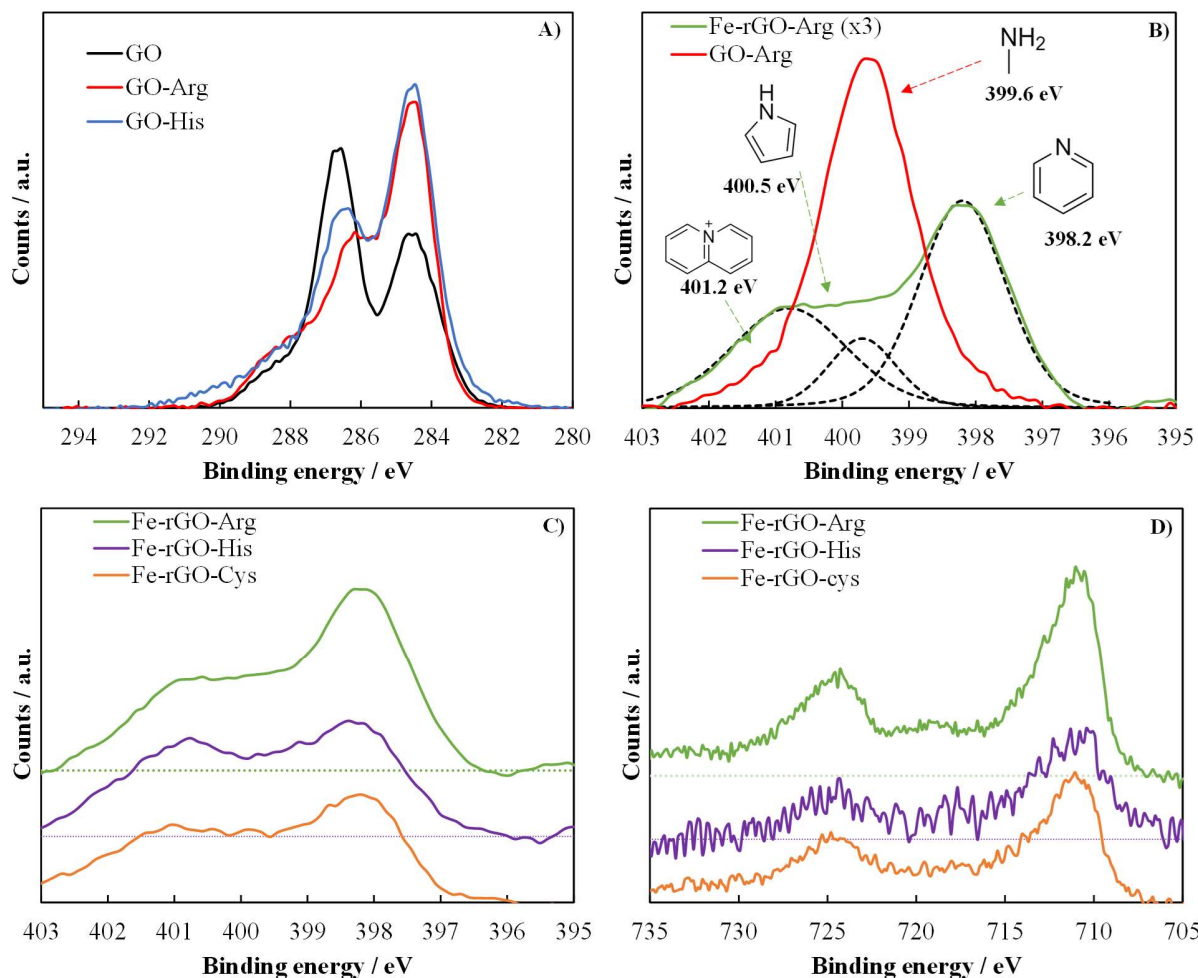
307 XPS analysis was used to investigate the composition and oxidation state of the synthesized  
308 samples. Table 3 collects the chemical composition (wt %) of the parent GO-AA and the derived  
309 Fe-rGO-AA catalysts. The formation of rGO is confirmed by the increase of carbon amount after  
310 the thermal treatment. The thermal reduction of surface oxygen groups is behind the decrease in  
311 O amount after pyrolysis, even though some oxygen remains forming the iron spinel detected by

312 XPS and XRD (see section 3.3). Comparison of the C1s profiles of GO, GO-His and GO-Arg,  
313 Figure 3.a, confirms a stronger reduction of GO after the functionalization of GO with arginine,  
314 as pointed out by the larger decrease in the band at 286.5 eV, which is related to the presence of  
315 labile epoxides of GO. Similar trend can be observed in the C1s spectrum of GO-Cys, Figure S1.

**Table 3.** XPS composition of electrocatalysts (wt.%)

sample	C	N	O	Fe	S
Fe-rGO-Arg	76.7	2.0	12.7	8.6	0.0
Fe-rGO-Cys	85.2	2.1	6.3	5.1	1.2
Fe-rGO-His	92.4	1.3	5.5	0.9	0.0

316  
317 The N-doping of r-GO is confirmed by the presence of nitrogen even after the thermal  
318 pyrolysis of the GO-AA in the presence of iron nitrate, observed in Table 1. The amount of  
319 nitrogen ranges from 1.3 to 2.1%, Table 3. Given the higher nitrogen content in the molecular  
320 structure of arginine, the maximum N loading is achieved in Fe-rGO-Arg. In the light of the low  
321 amount of nitrogen generated in Fe-rGO-His and the lower decrease in oxygen content observed  
322 in ultimate analyses, it is proposed that the imidazole group in histidine is hindering both the  
323 chemical reduction capacity of the amino function and its ability to form stable covalent bond  
324 with the surface oxygen groups of GO during the functionalization stage.



325  
 326 **Figure 3.** XPS of A) C1s region for GO, GO-His and GO-Arg, B) N1s region of GO-Arg and  
 327 Fe-rGO-Arg, C) N1s and D) Fe2p region of Fe-rGO-AA electrocatalysts.

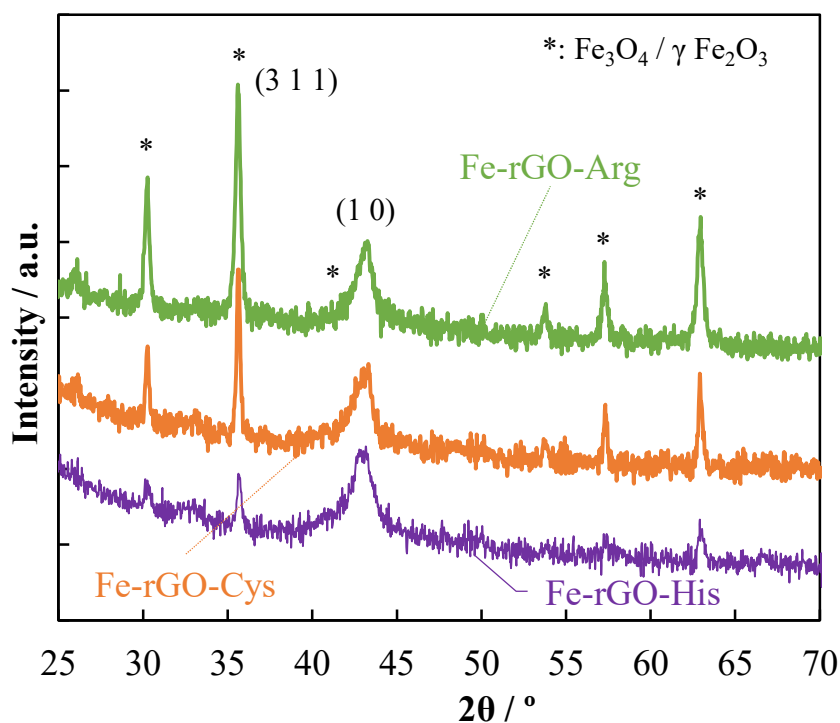
328 The cysteine functionalization of GO also delivers sulfur doping after the pyrolysis of the  
 329 sample, with Fe-rGO-Cys showing 1.2% of S on its surface. Binding energy of 163.9 eV in the  
 330 S2p peak of this sample (not shown for brevity sake) confirmed the formation of thiol groups  
 331 during the pyrolysis of this sample, while the formation of metal sulfides (binding energy of  
 332 161.5 eV) can be discarded [50]. Interestingly, the amount of iron detected by XPS on the  
 333 surface of rGO depends on the amino acid utilized for the functionalization. Thus, GO-Arg  
 334 seems to retain the highest amount of iron, 8.6%, whereas GO-His is barely able to fix iron on its

335 surface. The presence of an additional available amine group on arginine, and the sulfide group  
336 in cysteine could enhance the fixation of iron, since both groups are reported to interact with iron  
337 cations [51].

338 The changes in the chemical speciation and oxidation state of nitrogen species between Fe-Arg  
339 and Fe-rGO-Arg are compared in Figure 3.b, revealing that the amino functional group of the  
340 parent Fe-Arg sample are incorporated after pyrolysis in Fe-rGO-Arg, mainly as pyridine groups  
341 (N-6, 55%) and quaternary nitrogen (N-Q, 30%) and also, though in lesser amounts, as pyrrole  
342 groups (N-5, 15%) [52]. The nitrogen doping of the graphene layer was also observed in the  
343 other electrocatalysts, as evidenced by the XPS profiles in the N1s region shown in Figure 3.c.  
344 However, the distribution of nitrogen functional groups is different, being slightly richer in  
345 pyridines when cysteine is used as nitrogen source (45% N-6, 30% N-Q, 25% N-5), and richer  
346 in quaternary nitrogen for histidine (40% N-Q, 30% N-6, 30% N-5). In the case of the iron  
347 speciation, the Fe2p region revealed minor differences between the analyzed electrocatalysts,  
348 Figure 3.d. The observed peaks at 710.7 eV (Fe 2p<sub>3/2</sub>) and 724.2 eV (2p<sub>1/2</sub>) were located at  
349 intermediate values between those expected to Fe<sup>3+</sup> (711/724.8 eV) and Fe<sup>2+</sup> (709/722.6 eV), and  
350 the lack of the corresponding satellite peak in the 714-718 eV region (indicative of the presence  
351 of Fe<sub>2</sub>O<sub>3</sub>), can be considered as an evidence of the formation of Fe<sub>3</sub>O<sub>4</sub> [53], although the  
352 presence of small amount of Fe<sub>2</sub>O<sub>3</sub> cannot be discarded attending solely to this technique.

353 Figure 4 shows the XRD patterns for Fe-rGO-Arg, Fe-rGO-His and Fe-rGO-cysteine samples.  
354 Apart from the diffraction 10 peak from the reduced graphene oxide at 43.1 °, which indicates a  
355 short range order of stacked graphene layers [54], the main crystalline peaks observed in these  
356 patterns (marked as \*) are in good agreement with the reference data of the cubic spinel structure  
357 of Fe<sub>3</sub>O<sub>4</sub> (JCPDS card 65-3107), as expected from the XPS results. However, the reference

358 pattern of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS card 39-1356) is known to share the same features, hindering the  
359 accurate identification of the crystalline phases. The 311 peak in Fe-rGO-His and Fe-rGO-Cys is  
360 located at 35.53 °, which is between that of Fe<sub>3</sub>O<sub>4</sub> (35.42 °), and  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> (35.86 °). Therefore, it  
361 is proposed that a mixture of both crystalline phases, richer in Fe<sub>3</sub>O<sub>4</sub>, is likely to coexist in the  
362 surface of these samples [55]. In the case of the Fe-rGO-Arg, the position is shifted to about  
363 35.63 ° and the width of the crystalline peaks are larger, probably as a consequence of a larger  
364 contribution of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> phase in this sample. As for the dispersion of the iron, the intensity of the  
365 peaks follows the order Arg>Cys>His. It seems like the higher iron content of Fe-rGO-Arg  
366 favors the formation of larger crystalline domains, while the functionalization with histidine  
367 enhances the dispersion of iron on the surface of the obtained electrocatalyst.

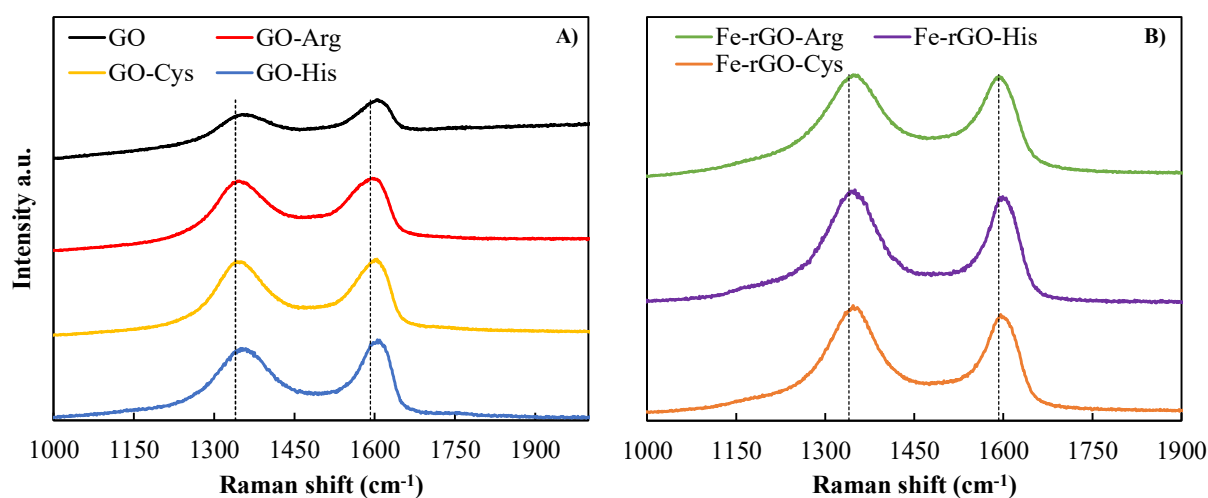


368

369

**Figure 4.** XRD patterns of Fe-rGO-Arg, Fe-rGO-His and Fe-rGO-Cys samples.

370 Raman spectroscopy was utilized to evaluate the structure of GO-AA and Fe-rGO-AA  
371 catalysts. First order Raman region is presented in Figure 5, where the peaks traditionally related  
372 to defect peak (D band) and graphitic lattice peak (G band) can be clearly seen [56]. The position  
373 of  $E_D$  and  $E_{2g}$  bands for graphite are also included as a guideline of the evolution of peak shift  
374 after functionalization and pyrolysis. The GO spectrum shows a tilted baseline owing to the high  
375 fluorescence of the sample [57], a feature that is lost after functionalization of GO with AAs due  
376 to the removal of oxygen groups, Figure 5.a. After the functionalization of graphene oxide, the G  
377 peak is shifted to lower Raman shift compared to that for GO, in agreement with previous reports  
378 regarding the structure of graphene oxide [58], which is attributed to the recovery of the  
379 hexagonal network of carbon atoms. In addition, a decrease in the FWHM of the D peak is also  
380 noticeable owing to the increase of the average size of  $sp_2$  clusters [59]. This decrease is related  
381 to the chemical reduction of the surface oxygen groups by the amino acids, a feature already  
382 reported in the literature [39]. All these trends are magnified after the pyrolysis of GO-AA in  
383 presence of iron nitrate, Fig. 5b.



384

385 **Figure 5.** Raman spectra of A) GO-AA samples and b) Fe-rGO-AA electrocatalysts.

386 As reported by Ferrari et al., the Tuinstra-Koenig relationship between  $I_G/I_D$  and the lateral  
 387 crystallite size cannot be applied in short-range ordered materials [60], explaining why this ratio  
 388 decreases after the functionalization and thermal treatments of GO, in spite of the strong removal  
 389 of oxygen functionalities observed by ultimate analyses, TPD and XPS. Moreover, it is difficult  
 390 to spot differences between the Raman spectra of GO-AA and Fe-rGO-AA. Consequently, all the  
 391 spectra were deconvoluted using 5 peaks (G, D, D', D'' and D\*) according to the  
 392 recommendations made for the analysis of Raman spectrum of GO and rGO [58], and the most  
 393 relevant parameters are compared in Table 4.

**Table 4.** Deconvolution parameters of Raman spectra

Sample	$F_{D''}$ / $\text{cm}^{-1}$	$I_{D^*}$ / %	$I_D/(I_D+I_G)$ / %
GO-Arg	183.1	1.4	81.3
GO-Cys	187.8	0.8	82.0
GO-His	160.7	1.5	84.4
Fe-rGO-Arg	140.6	4.5	71.8
Fe-rGO-Cys	137.3	3.4	72.9
Fe-rGO-His	114.3	4.4	72.0

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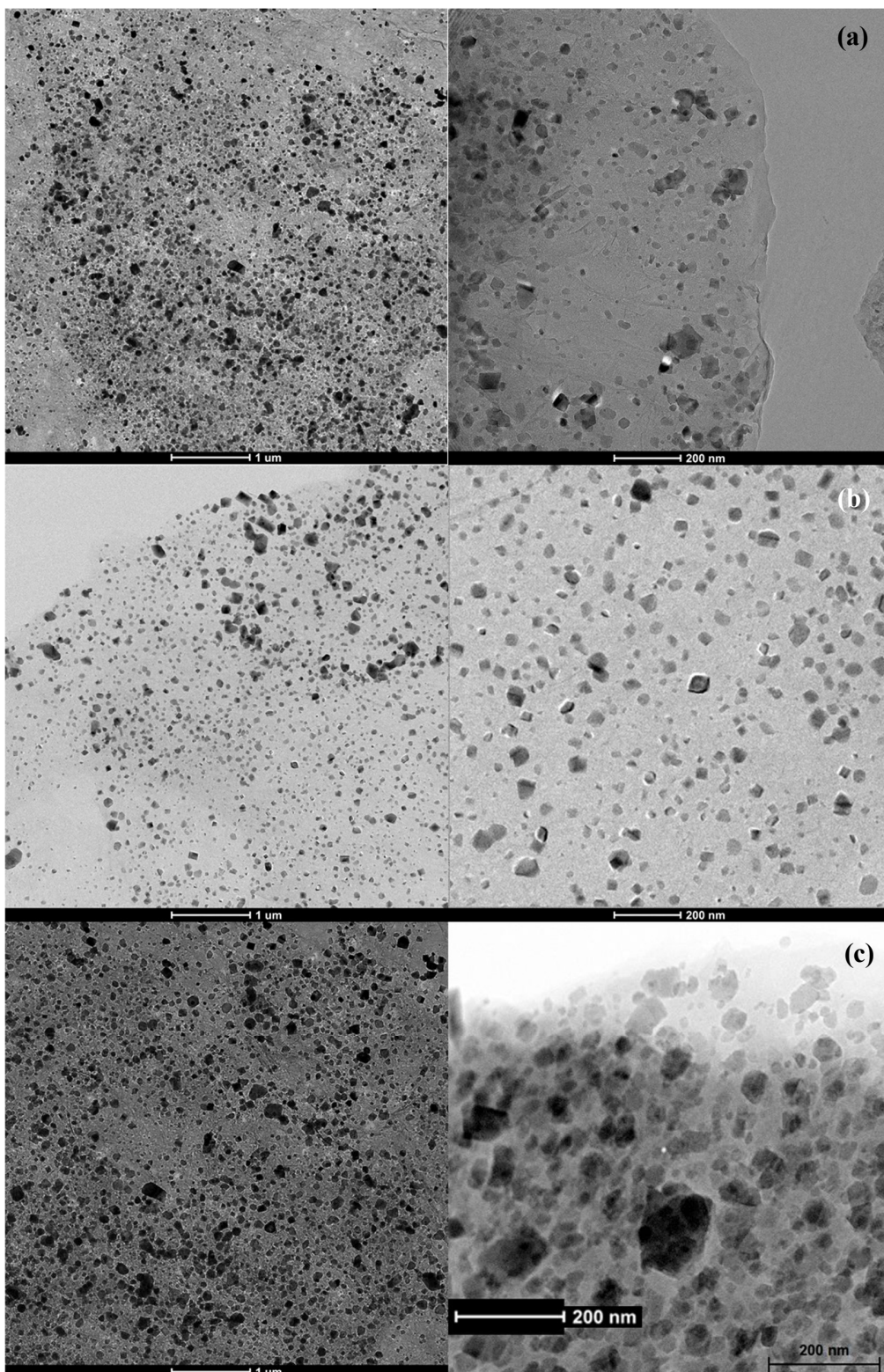
395 It is seen that the narrowing of the FWHM of D'' band,  $F_{D''}$ , is stronger in the case of  
 396 histidine, followed by arginine and cysteine. According to the interpretation given by Vollebregt  
 397 et al., this band is related to the presence of amorphous phase [61], and therefore  
 398 functionalization with histidine seems to deliver the higher removal of disordered carbon. As  
 399 expected, the Fe-rGO-His sample shows even lower values of this parameter, since the pyrolysis  
 400 at 800 °C removes preferentially the less stable amorphous carbon content. A different trend is

401 observed in the intensity of D\* band,  $I_{D^*}$ , which is related to the presence of  $sp_3$  dangling bonds  
402 as those found in carboxyl groups and also as edges and in-plane defects [58]. The larger  
403 intensity of this band after pyrolysis can be explained due to the removal of surface oxygen  
404 groups, which generates carbon free sites at edge positions, and due to the incorporation of  
405 nitrogen groups as pyrroles, which are known to generate surface defects because of the  
406 reconstruction of the edge carbon lattice [62]. Moreover, TPD results have showed that amino  
407 acid functionalization removes carboxylic moieties, Figure 2, which is in good agreement with  
408 the low intensity value of  $I_{D^*}$  obtained for GO-AA samples.

409 As for the lateral crystalline size,  $L_a$ , Cuesta et al. proposed the  $I_D/(I_D+I_G)$  ratio as a more  
410 adequate alternative to the  $I_D/I_G$  Tuinstra-Koenig ratio for the evaluation of the Raman parameter  
411 of low structural order carbon materials [63]. Indeed, the modified  $I_D/(I_D+I_G)$  ratio has been  
412 recently verified as a better descriptor of the inverse of the crystalline size in the basal plane  
413 direction,  $1/L_a$ , for GO and rGO [58] Table 4 shows that the values of  $I_D/(I_D+I_G)$  for the GO-AA  
414 samples are very similar no matter the amino acid used (81.3-84.4%). However, the pyrolysis in  
415 the presence of iron nitrate lowers this value by an average of 10%, Table 4. The increase of the  
416  $sp_2$  domain sizes (e.g increase of  $L_a$ ) after pyrolysis is a positive change, since it has been related  
417 to improved electrical conductivity [59], a necessary feature for electrocatalysts.

418 To probe the microstructure and morphology of the synthesized samples, TEM images of Fe-  
419 rGO-AA catalysts were taken at different magnifications, Figure 6. A common point in these  
420 three images is the presence of metal nanoparticles in the form of amorphous and highly dense  
421 particles on top of the wrinkle and much less dense structure of the reduced graphene oxide. The  
422 amount and agglomeration of metal particles detected at low magnification (left panel in Figure  
423 6) is much higher for Fe-rGO-Cys and Fe-rGO-Arg, in accordance with the metal content

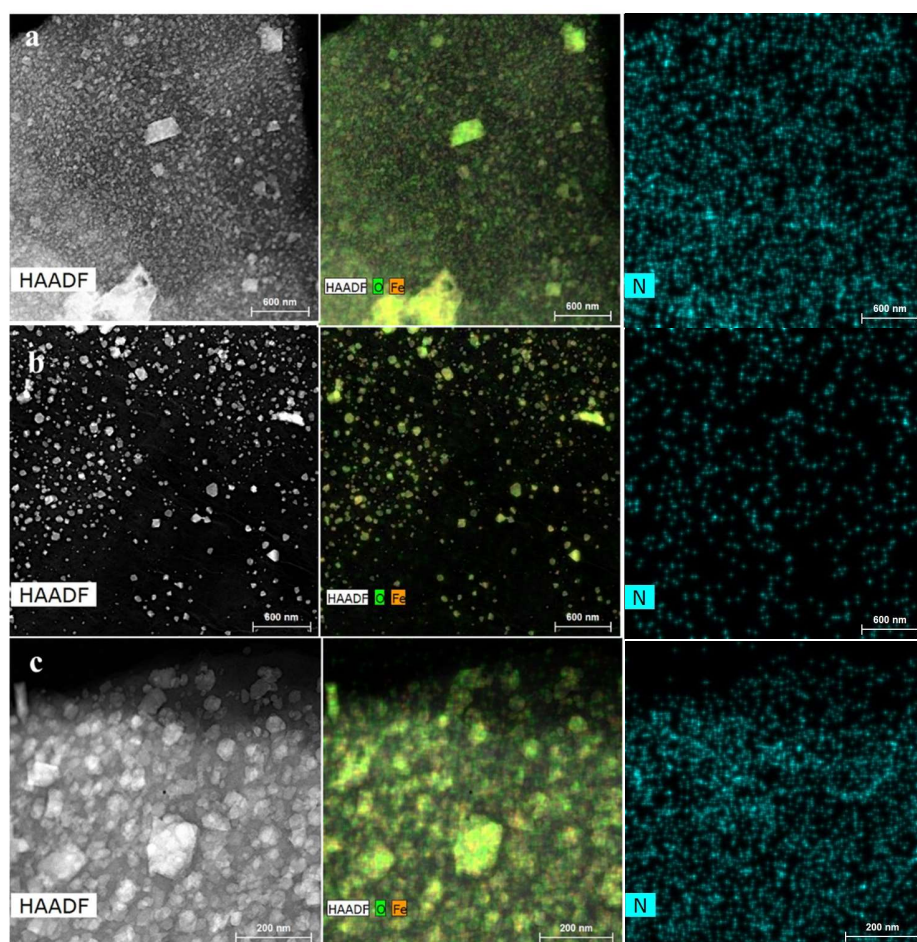
424 detected by XPS and the presence of crystalline phases on XRD. Conversely, smaller size  
425 nanoparticles can be observed in these samples at higher magnification (right panel in Figure 6),  
426 especially in the case of Fe-rGO-His. Thus, average particle size of ca. 25 nm, 30 nm and 40 nm  
427 were determined for samples prepared using arginine, histidine and cysteine, respectively.



428

429 **Figure 6.** TEM images of (a) Fe-rGO-Arg, (b) Fe-rGO-His and (c) Fe-rGO-Cys

430 Likewise, elemental mapping of the samples is shown in the Fig.7. As can be seen, the flakes  
431 are mainly composed by carbon, whereas the black spots are rich in iron and oxygen, in  
432 agreement with the detection of iron oxides by XRD. Differently, nitrogen have a uniform  
433 distribution in the structure for all the samples, which is probably generated by the homogeneous  
434 functionalization of GO with AA. EDX mapping confirms that the iron distribution is greatly  
435 enhanced in Fe-rGO-His sample, showing a decreased crystalline size and the absence of  
436 massive crystalline clusters. According to these maps, it can be concluded that nitrogen  
437 incorporation into rGO, and the addition of iron to the catalysts, have been successfully obtained.



438  
439 **Figure 7.** HAADF + EDX images of (a) Fe-rGO-Arg, (b) Fe-rGO-His and (c) Fe-rGO-Cys

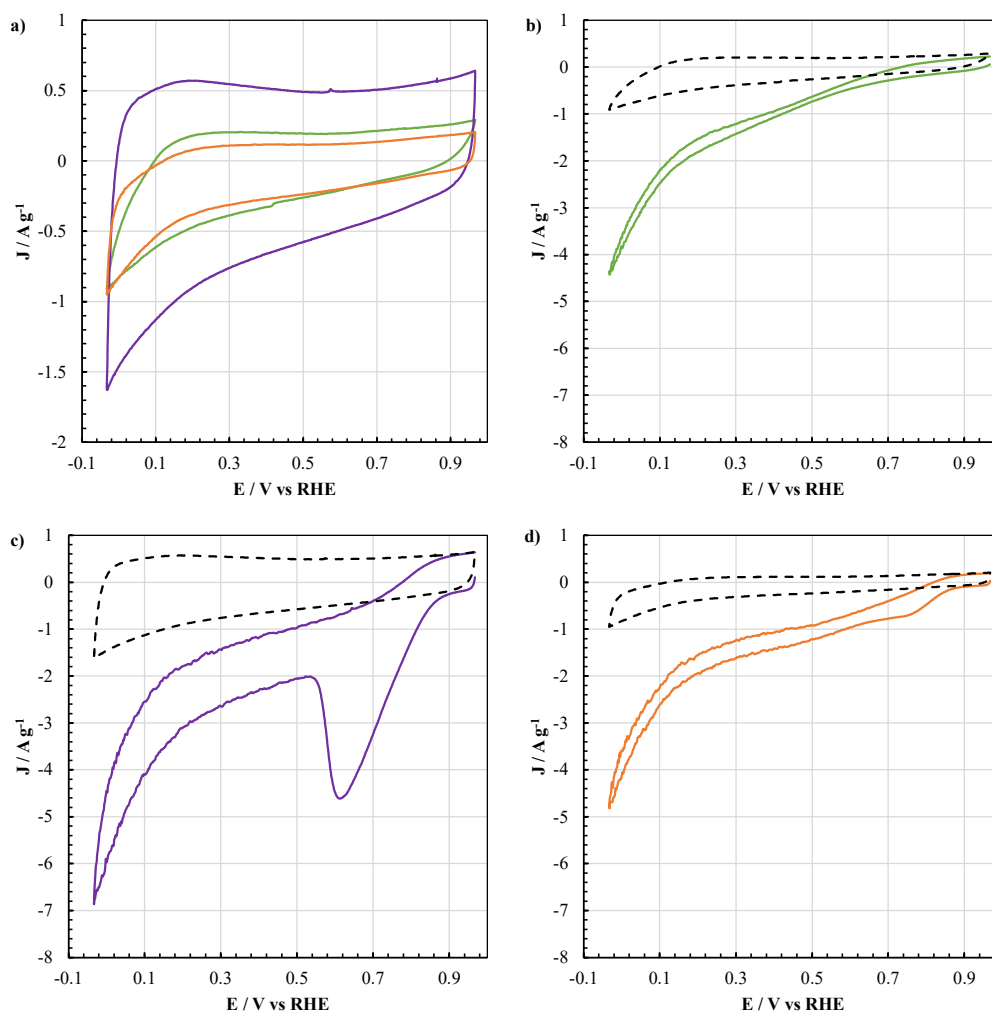
### 440 3.3. Electrochemical characterization of Fe-rGO-AA electrocatalysts

441 Cyclic voltammetry (CV) between 0 and 1 V vs RHE of all the iron-containing samples in 0.1  
442 M KOH were measured at 10 mV/s under inert and oxygen-saturated atmosphere using a 3-  
443 electrode cell study, and the results are shown in Fig. 8. The potential window was set in order to  
444 clarify the onset potentials of the reduction currents of oxygen reduction reaction. The CV  
445 response of all the three samples in absence of oxygen mainly presents a capacitive behavior.  
446 The negative polarization of the Fe-rGo-His and Fe-rGO-Arg electrodes gives rise to a small  
447 cathodic current at potential lower than 0.1 V, Figure 8.a. This current is probably related to the  
448 reduction processes observed in alkaline media for Fe<sub>3</sub>O<sub>4</sub> electrodes, which are reported to  
449 happen around that potential value [64]. None of the other known redox features of Fe<sub>3</sub>O<sub>4</sub> and  
450 Fe<sub>2</sub>O<sub>3</sub> species are observed in these analyses, since they are likely to happen outside the potential  
451 limits of the CV.

452 The double layer capacitance of these samples was estimated from the area inside the CV  
453 recorded in inert atmosphere, obtaining values of 56 F/g for Fe-rGO-His, 22 F/g for Fe-rGO-Arg  
454 and 17 F/g for Fe-rGO-Cys. The capacitance drop in the latter samples is probably related to the  
455 massive deposition of iron oxide nanoparticles on top of the surface of rGO, disabling a large  
456 fraction of the surface area for the generation of the electric double layer. Differently, Fe-rGO-  
457 His presents not only larger capacitance, but it also shows a well-defined, box-shaped CV,  
458 pointing out the higher surface area and excellent electrical conductivity of this electrode, both  
459 valuable features for the performance of electrocatalysts. The improved conductivity is probably  
460 related to the higher structural order observed in Raman (Figure 5 and Table 4) and to the  
461 preferential N-Q doping of the sample confirmed by XPS (Figure 3), which has been reported to  
462 enhance the electrical conductivity of N-doped carbon materials [65]. Moreover, the lower

463 amount of iron observed in Fe-rGO-His, along with the higher double layer capacitance  
464 contribution, explain why the cathodic current of Fe<sub>3</sub>O<sub>4</sub> reduction is barely seen. However, the  
465 slope of the negative scan on this electrode also increases at 0.1 V, pointing out some small  
466 contribution of the reduction of iron species observed in Fe-rGO-His sample.

467 Further comparison of the CVs recorded in O<sub>2</sub>-free and O<sub>2</sub>-saturated electrolyte are delivered  
468 in Figure 8.b-d for a better assessment of the ORR activity in these samples. In oxygen-saturated  
469 electrodes, strong and clear cathodic currents are observed in all the cases, which are  
470 undoubtedly related to the electrochemical reduction of oxygen. Nevertheless, the onset potential  
471 of oxygen reduction reaction depends on the electrocatalyst of choice, following the order Fe-  
472 rGO-His >  
473 Fe-rGO-Cys >> Fe-rGO-Arg. Thus, the catalytic activity is clearly unrelated to the amount of  
474 iron in the electrocatalysts, Table 2. A result that agrees with the studies of Lefèvre and Dodelet,  
475 who reported that Fe-N electrocatalysts prepared by pyrolysis of iron porphyrin had iron contents  
476 as low as 0.5 wt.%, showing analogous ORR activity than similar samples prepared with higher  
477 loadings [66]. In fact, the low electrical conductivity of iron oxide can explain the higher  
478 electrode resistance of Fe-rGO-Arg, as depicted by the tilted CV when compared to the square-  
479 shaped CVs recorded in N<sub>2</sub> atmosphere for the other electrocatalysts, Figure 8.a. Even though  
480 cathodic currents of Fe-rGO-His at medium and low potentials are the largest ones, the similar  
481 ORR onset potentials registered for Fe-rGO-His (0.88 eV) and Fe-rGO-Cys (0.86 eV) do not  
482 allow to conclude the superior performance of the former sample.



483

484

485 **Figure 8.** A) Cyclic voltammetric measurements of Fe-rGO-AA samples in inert atmosphere.

486 Comparison of CV recorded in inert (dotted lines) and oxygen saturated (solid lines) 0.1 M KOH

487 electrolyte for B) Fe-rGO-Arg C), Fe-rGO-His and D) Fe-rGO-Cys samples.

488

489 The Rotatory Disk Electrode (RDE) assembly was used to study the ORR activity of these

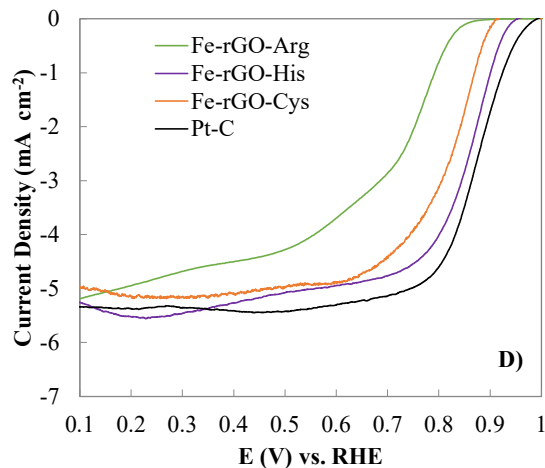
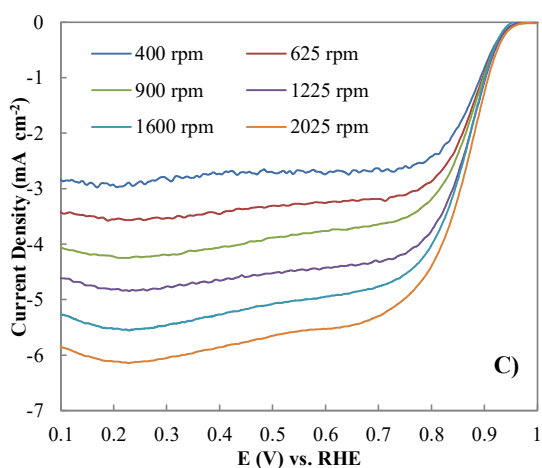
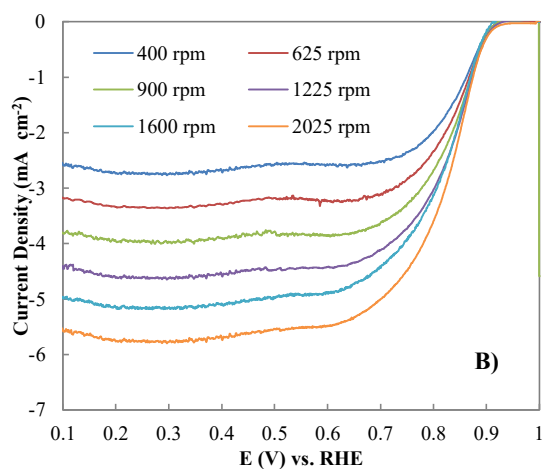
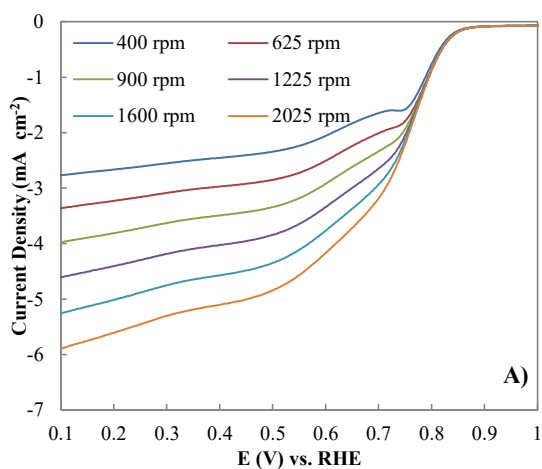
490 samples. Linear sweep voltammetry (LSV) curves at different rotation ratios were determined for

491 all three samples and are shown in Figure 9. As expected, all the catalysts show increased ORR

492 limiting current as rotation rate increases. However, the shape of the profile is different

493 depending on the sample. In accordance with the three-electrode cell study, Figure 8, the catalyst  
494 showing the lowest activity is Fe-rGO-Arg. It shows a 2-wave profile, with the second one  
495 starting at ca. 0.4 V. This feature has been ascribed to a mixed 2+2 e<sup>-</sup> ORR mechanism [67],  
496 where oxygen is reduced at high potentials to hydrogen peroxide, delivering 2 e<sup>-</sup>, while hydrogen  
497 peroxide is finally reduced to water at medium and low potentials (2<sup>nd</sup> wave of the LSV profile),  
498 producing the final 2 e<sup>-</sup>. Differently, Fe-rGO-Cys and Fe-rGO-His presents only one wave,  
499 reaching their diffusion limiting currents at potentials ca. 0.6 V. The I-V curve in the transition  
500 region of these samples is steeper than in the case of Fe-rGO-Arg, pointing out the lower  
501 resistance of these electrodes. Although the shape of the LSV profiles of the two latter samples  
502 are somehow similar, Fe-rGO-His shows a higher onset potential, indicating the improved  
503 catalytic activity of this electrocatalyst.

504



505

506

507 **Figure 9.** LSV curves of A) Fe-rGO-Arg, B) Fe-rGO-Cys, and C) Fe-rGO-His catalysts. D)

508

comparison between LSV profiles at 1600 rpm

509 For a clearer assessment of the activity, the LSV profiles of all the catalysts have been

510 gathered in Figure 9.d and compared with a commercial catalyst sample (20% platinum on

511 Vulcan XC72 carbon black, labelled as Pt-C). The ORR factors related to catalytic performance,

512 as the onset potential ( $E_0$ , determined at  $0.1 \text{ mA/cm}^2$ ), the half wave potential ( $E_{1/2}$ , determined at

513  $2.6 \text{ mA/cm}^2$  for all the samples) and the electron transfer numbers ( $n$ , calculated from the slope

514 of the KL plot as detailed in the experimental section), were determined from these LSV profiles

515 and compared in Table 5. Since the number of electrons is best determined by the use of a  
516 rotating ring disk electrode, the results of the KL analysis of PtC are also included for  
517 comparison purposes. It is now evident that the onset potential, a vital indicator for ORR activity,  
518 follows the order Fe-rGO-His>Fe-rGO-Cys>Fe-rGO-Arg. The superior activity of Fe-rGO-His is  
519 maintained at the half-wave potential. Moreover, the Koutecky Levich plots showed that oxygen  
520 is mainly reduced following the four-electron transfer pathway in Fe-rGO-His ( $n=3.94$ ) and Fe-  
521 rGO-Cys ( $n=3.89$ ), while ORR proceeds through a mixed  $2+2 e^-$  mechanism ( $n=2.82$ ) in Fe-  
522 rGO-Arg. The ORR activity of Fe-rGO-His greatly improves that achieved in metal free amino  
523 acid-doped rGO [17, 18, 30]. Even though Fe-rGO-His shows a lower amount of iron and  
524 nitrogen than the other catalysts, the lack of relationship between N and metal content has  
525 already been exposed in the literature [23, 25, 68]. To our knowledge, there are not works about  
526 the preparation of Fe-N-rGO catalysts by pyrolysis of iron salts and amino acids in the presence  
527 of GO, therefore further activity comparisons are done against Fe-N-C catalysts prepared using  
528 different supports. In this sense, activity of Fe-rGO-His is at least comparable in terms of  $E_{on}$  and  
529  $E_{1/2}$  to that of: the Fe-N-C catalysts prepared from pyrolysis of hydrothermal carbons doped with  
530 histidine and iron nitride and etched with ammonia [30]; Fe-N doped template carbons obtained  
531 by hard template method using histidine and iron (III) mixtures [28]; and doped mesoporous  
532 carbon nanostructures prepared from cysteine and porphyrins using SBA-15 as hard template  
533 [29]. Even though the commercial Pt-C catalyst still outperforms the histidine-based  
534 electrocatalyst by ca. 30 mV, Table 5, the scarcity of platinum, the low cost and high availability  
535 of iron and amino acids and the better performance in terms of resistance to cell-crossover or  
536 poisoning in Fe-N catalysts could pay off for the slightly lower activity [69, 70].

537

538

**Table 5.** ORR performance parameters

	$E_{0.1}$	$E_{1/2}$	$n_{0.65}$
Sample	V	V	--
Pt-C	0.98	0.88	3.96
Fe-rGO-Cys	0.91	0.80	3.89
Fe-rGO-Arg	0.85	0.72	2.82
Fe-rGO-His	0.94	0.85	3.94

539

## 540 **Conclusions**

541 Graphene-based electrocatalysts were prepared by pyrolysis of graphene oxide functionalized  
542 with three different amino acids, arginine, cysteine and histidine, in the presence of Fe(NO<sub>3</sub>).  
543 Amino acid functionalization rendered a decrease of oxygen content due to the chemical  
544 reduction of GO, following the order arginine > cysteine > histidine. The chemical reduction of  
545 functionalized GO and the preferential functionalization via amide bond formation was  
546 confirmed by FTIR and TPD measurements. The changes in the S:N atomic ratio for cysteine  
547 and the absence of CO<sub>2</sub> evolution during TPD of GO-AA samples revealed that amino acids are  
548 partially decomposed during the functionalization. Elemental and XPS analyses showed the  
549 formation of nitrogen surface groups after the pyrolysis of GO-AA, in agreement with previous  
550 results about the preparation of N-doped reduced graphene oxide following this protocol.  
551 However, XPS revealed that, when iron nitrate is present during pyrolysis, the level of N-doping  
552 decreases, while the carbon content and graphitization degree increased for all the obtained Fe-

553 rGO-AA samples, as observed by Raman. The combined results of XPS and XRD suggested the  
554 formation of crystalline mixtures of iron spinel and iron oxide species in the catalysts, especially  
555 for cysteine- and arginine-based samples, which had higher iron loadings. TEM/EDX analyses  
556 also confirmed a lower amount and better dispersion of iron nanoparticles on histidine-based  
557 electrocatalyst.

558 Furthermore, the electrochemical characterization in 0.1 M KOH showed that the  
559 electrocatalysts prepared using cysteine and histidine delivered a larger catalytic activity for  
560 ORR and an improved charge transfer, whereas the excessive fixation of iron oxide in arginine-  
561 based electrocatalyst seems to be detrimental due to the negative impact in electrical resistance,  
562 hindering the performance as ORR catalyst of this sample. RDE measurements revealed that Fe-  
563 rGO-His was able to catalyze ORR following the desirable 4 e<sup>-</sup> pathway, achieving onset  
564 potentials only 40 mV lower than commercial platinum-based catalysts. These promising results  
565 confirm that nanostructured Fe-N reduced graphene oxide catalysts with adequate ORR activity  
566 can be prepared using amino acids as sustainable and low-cost precursors.

567

## 568 **Acknowledges**

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570

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