

1 **Wavelength and energy dependence on ablation dynamics under femtosecond laser pulses**
2 **observed by time-resolved pump–probe microscopy**

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6 **ABSTRACT**

7 Pump–probe femtosecond microscopy has been applied to reveal the time evolution of
8 the ablation processes at sub-picosecond resolution with combined pulses of different
9 photon energies. Experimental results revealed that differences in the photon energies of
10 ablation pulses influenced the temporal response of the material. Significant differences
11 in the excitation and melting of the material within 10 ps of irradiation were observed.
12 The analysis of dynamic Newton’s rings observed with the pump–probe technique
13 confirmed and explained the faster expansion of ablated matter when using shorter
14 irradiation wavelengths. The results also implied that optical properties of the material
15 play a significant role.

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18 INTRODUCTION

19 Laser-based techniques allow the modification of materials in the melting,
20 spallation, and ablation regimes, which includes the gentle removal of just a few atoms
21 or molecules to a high material mass removal rates. Several different analysis techniques
22 have been used in an effort to visualize and understand the phenomena involved in the
23 interaction of a femtosecond laser pulse with solid matter from surface emission up to
24 permanent damage. Depending on the intended application of laser ablation, different
25 characteristics of the laser pulse will be of more interest. Various studies over the last
26 decades have investigated the influence of the pulse width (1-5). Ablation with
27 nanosecond lasers is dominated by laser–matter and laser–plasma processes, as the pulse
28 duration is long enough that it persists as evaporation of material is taking place. These
29 processes affect the total deposition of energy on the target, which influences the resulting
30 crater and damage formation (6). The key difference with femtosecond pulses is that the
31 laser–matter interaction is shorter in time than all the processes induced by this interaction
32 which, among other advantages, results in material being removed only from a confined
33 region corresponding to the irradiated area (5, 7-8).

34 The influence of the laser wavelength on the properties of the plasmas produced
35 when using short pulses to irradiate the target (9-14), have been studied from the point of
36 view of the acquisition of LIBS spectra. In the case of ultrashort pulses, different
37 experimental techniques including filamentation LIBS (15-17), ICP (18), and
38 micromachining and drilling (12) have been applied to investigate the role of this
39 parameter. Different analysis techniques can help reveal the processes dominant during
40 ablation; for example, by the application of fast image acquisition, as shadowgraphy, or
41 the use of streak cameras (19-21). Although these visualize the processes that induce
42 ablation, they do not show what is happening in the structure of the material. Pump–probe
43 microscopy has proved to be a useful and robust tool to monitor phase-change processes
44 (22-28). The use of femtosecond laser pulses for both ablating and probing the irradiated
45 target offers a temporal resolution limited by the pulse width, providing deeper insight
46 into the processes involved in ablation and plasma formation. Additionally, the absence
47 of laser–plasma interaction when using femtosecond pulses, makes it possible to deposit
48 more laser energy into the target compared to longer laser pulses (29,30). The process
49 occurring on dielectrics are complex and demand high energy densities to occur (31-33)
50 while on metals, due to the absence of a band gap in their electronic configuration, the

51 main mechanism of energy absorption is by free carriers, so less energy is required from
52 the laser pulse to induce the ablation (29, 34-36).

53 The present study applies time-resolved pump–probe microscopy to determine the
54 effect of irradiation wavelength on the processes taking place prior to plasma formation
55 during femtosecond laser ablation. Time-resolved pump–probe microscopy provides
56 insight into the morphological dynamics of the material with high temporal resolution.
57 Gold has been used as test sample due to its significant different reflectance behavior
58 along the UV-VIS spectra and the easiness in the fabrication of smooth thin layers with
59 minimum scattering effect due to sample roughness. Relevant differences, even at sub-
60 picosecond times, are reported, highlighting the importance of the optical properties of
61 the material.

62

63 EXPERIMENTAL SET-UP

64 The experimental set-up has been described in detail in previous references
65 (27,28). Briefly, it consists of an regeneratively-amplified Ti:Sa laser providing p-
66 polarized 35 fs FWHM pulses at 800 nm with a mean energy of 3.5 mJ. By means of a
67 80/20 beam splitter and two optical delay lines, two independent laser branches (pump
68 and probe) arrive the sample at selectable times. The temporal precision that can be
69 achieved with these two stages is 6.67 fs, so the temporal resolution of the experiment is
70 ultimately limited by the length of the laser pulses. The energy of the pump beam and
71 probe beams are controlled by optical attenuator, and pulses are focused onto the target
72 by a plano-convex lens with a 150 mm focal length. In all the circumstances, the probe
73 beam is adjusted below the ablation threshold. An electromechanical shutter ensures that
74 only a single pump pulse irradiates the target for each event. A harmonic generator allows
75 the transformation of an incoming beam at 800 nm into two beams at 400 nm and 266
76 nm.

77 Light reflected by the target passes back through the microscope objective and is
78 reflected by a dichroic mirror and finally focused onto a 10-bit CMOS camera
79 synchronized (As the shutter) by the laser amplifier. Specific dichroic mirrors, bandpass
80 filters and polarizing elements were added for the different combinations of 800, 400 and
81 266 nm wavelengths for the pump and probe beams as detailed in Figure 1.

82 The samples studied in the present work were 200 nm thick gold films deposited
83 on glass substrates by DC sputtering. The roughness of the films was lower than 5 nm, as
84 measured by atomic force microscopy. A fresh part of the sample surface was used for

85 each laser event. Three images of the surface were acquired for every event: the first
86 image recorded the sample surface prior to irradiation; the second image corresponds to
87 the arrival of the probe beam with a predetermined delay relative to the pump beam, and
88 captures the process as it takes place; the third image is taken several seconds after
89 irradiation, and shows the permanent damage to the sample surface. Images
90 corresponding to each laser shot were processed using a custom MATLAB script, which
91 displays the first and second images and allowing the user to select the area of interest
92 where ablation is observed. The script then outputs the normalized reflectivity.

93

94 RESULTS AND DISCUSSION

95 Three main stages can generally be distinguished during femtosecond laser ablation,
96 although these may be initiated by different mechanisms depending on the nature of the
97 target (29, 37-39). The first process is excitation, which is characterized by a bright area
98 where the laser pulse impinges on the sample. In this stage, lattice electrons are excited
99 by the laser's electric field and promoted from the valence band to the conduction band
100 by multiphoton ionization or inverse Bremsstrahlung. This increases the free electron
101 population and can be observed as an increase in the reflectivity of the target.
102 Subsequently, after a few tens of picoseconds, target melting starts, accompanied by a
103 dramatic decrease of the reflectivity which manifests itself as an expanding dark area. In
104 contrast to excitation, this dark area grows radially outwards, as all electrons in the
105 irradiated area are excited by this time. This melting results from relaxation processes
106 such as electron-phonon collision, with different pathways active depending on the input
107 energy and the characteristics of the lattice (29, 39-40). These collision processes result
108 in a melting front propagating through the irradiated volume. Once the melting front
109 reaches the bulk-molten material interface, expansion begins. In this stage, under certain
110 conditions of the absence of roughness, ring patterns which evolve with time have been
111 observed (26, 27, 41, 42). These arise due to differences between the refractive indices of
112 the external layer, which remains in a solid-liquid state, and the vaporized material
113 contained between this layer and the non-ablated material.

114 Schematic side and top-down views of these processes are shown in Figure 2.; the top-
115 down views in the bottom row illustrate the changes in reflectivity produced by the
116 interaction of the probe beam with the target. During excitation, the increased electron
117 population at the surface causes an increase in the reflectivity, which remains until
118 electron de-excitation occurs. On the other hand, material melting, caused by de-

119 excitation of electrons by collision processes, is observed as a decrease in the reflectivity
120 of the sample, starting in the center of the irradiated area and growing radially. Finally,
121 once surface deformation starts, the outer surface of the target consists of a mixture of
122 solid and liquid phases encapsulating the vaporized material and resultant voids,
123 ultimately giving rise to the dynamic Newton's rings patterns.

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132 solid and liquid phases encapsulating the vaporized material and resultant voids,
133 ultimately giving rise to the dynamic Newton's rings patterns.

134 Figure 3 shows different time-resolved micrographs to compare the evolution of the
135 surface of irradiated material when using each of the three harmonics as pump irradiation.
136 The pump energy was fixed at 20 μJ , and 400 nm light was used as the probe beam.
137 Contrast and brightness have been enhanced identically in the three situations in order to
138 make the differences more visible. As observed, pump-probe time-resolved micrography
139 allows the visualization of the sequential excitation, melting and vaporization processes
140 and the differences derived on the photon energy to excite the sample. Thus, a local
141 increase of the reflectivity of the irradiated area is produced in the first picosecond after
142 irradiation for all experiments. The comparison of the micrographs at 2 ps reveals clearly
143 the larger reflectivity as the pump wavelength decreases.

144 Once the melting front reaches the interface with the bulk material, it is reflected and
145 the target starts to deform, as depicted in Figure 2. Target deformation produces a
146 multilayer system at the target surface, giving rise to Newton's rings. These start to
147 develop earlier (~ 1 ns) for shorter wavelengths, while are barely seen at 800 nm. The
148 transmission of more energy from electrons to phonons induces faster melting of the
149 material, forcing the vaporized material encapsulated by the target surface to expand
150 faster. The result is a more pronounced curvature of the surface dome yielding more
151 numerous and more closely-spaced the Newton's rings. At 800 nm, no evidence of
152 Newton's rings is observed in the interval studied.

153 A detailed intensity profile of the different micrographs covering the timescale from
154 time zero to 2 ns is found in Figure 4. The plot shows the normalized intensity as well as
155 the error bars for three replicated measurements processed with the MATLAB script
156 detailed in the experimental section. In all the cases, an identical central region on each
157 image was considered for computing.

158 For 266 nm excitation (solid triangles), it is quite evident the significant increase in
159 reflectivity observed up to 2 ps after irradiation, in consonance with the large target
160 excitation at such wavelength. After excitation, melting proceeds at fast rate due to the
161 high electron density which increases the rate of electron–phonon collisions. Similar
162 results are found at 400 nm (open squares), also exhibiting (although less pronounced) an
163 increase of reflectivity due to electron excitation, with a maximum reflectivity around 1
164 ps earlier than for the measurements using the 266 nm wavelength. The maximum
165 reflectivity was lower for 400 nm irradiation, explainable due to the lower photon energy.
166 These measurements were repeated at half the energy at both wavelengths without any
167 significant change in the behavior. In both cases, the error bars correspond to the
168 averaging of three replicated measurements. For 800 nm irradiation, it is quite evident the
169 plateau-like behavior of the surface reflectivity during the first 10 ps, confirming that
170 irradiation at longer wavelengths requires a higher number of photons to induce sample
171 excitation.

172 The role of the probe beam can't be ruled out in our time-resolved microscope. To
173 check its effect, an experiment was performed with permuted 400 and 800 nm beams as
174 pump and probe beams. This combination was chosen due to the large differences in
175 reflectivity of gold for such wavelength as can be seen in Figure 5, where the reflectivity
176 of Ag and Al are included for comparison. As observed, Au reflectivity at 800 nm is close
177 to 98% while at 400 nm is around 37%. Figure 6 shows the time evolution of the
178 normalized reflectivity using permuted 800/400 and 400/800 pump–probe configurations
179 at higher photon densities compared to the one used in Figure 4. As shown for the 800/400
180 pump/probe scheme (circles) an increase in the reflectivity is observed during excitation
181 at times before 1 ps, followed by a decrease in reflectivity associated with melting of the
182 target. At a difference of the results in Figure 4, the increased photon density has been
183 high enough to induce the ablation in the target. Although not calculated for this work, it
184 seems evident the existence of a wavelength-specific threshold energy value. The
185 dependence of such value with sample type or sample conditions are out of the scope of
186 this work, although a previous references (43) dealing with the calculation of plasma

187 ignition threshold fluence values by means of optical emission spectroscopy for different
188 metals revealed a dependence with the work function. The correlation between the plasma
189 formation values and our time-resolved reflectivity measurements is yet to be calculated
190 in future works. Concerning the 400/800 configuration (squared symbols in Figure 6),
191 despite the higher photon density, a plateau is observed, while in Figure 4 (with a 400/400
192 configuration and 20 μJ of pulse energy) a clear increase in reflectivity with a maximum
193 at 1 ps is observed. The difference in the results observed when using a 400 nm and 800
194 nm pump and probe respectively can be explained based on the natural reflectivity of the
195 material. The probe pulses reflect the transient processes taking place in the irradiated
196 material. Under 800 nm probe conditions, the native sample reflectivity will be that high
197 that any reflectivity increase will be barely detectable.

198

199 **Conclusions**

200 The effect of photon energy during femtosecond laser irradiation was studied using time-
201 resolved pump–probe microscopy. It was found that during the excitation stage, the pump
202 wavelength induces differences in excitation time, but that this is also influenced by the
203 optical properties of the material, as the penetration depth and native reflectivity of the
204 sample also play a role. Melting and expansion of the irradiated material were also
205 investigated by the analysis of Newton’s rings, but the most relevant parameters in these
206 stages proved to be the photon energy and photon density. The role of the probe
207 wavelength was also studied under different conditions. The higher reflectivity of gold at
208 800 nm means that an increase in reflectivity as the sample is excited by femtosecond
209 pulses is not observed, even when increasing pump pulse energy, although the Newton’s
210 rings grow faster. The latter demonstrates the faster expansion of the ablated material
211 when the material is irradiated using shorter wavelengths.

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221 **References**

- 222 1 Chichkov, B.N., Momma, C., Nolte, S. et al. Appl. Phys. A (1996) 63: 109.
223 Femtosecond, picosecond and nanosecond laser ablation of solids.
224 <https://doi.org/10.1007/BF01567637>
225
- 226 2 Ashkenasi, D., Rosenfeld, A. et al. Appl Phys A (1997) 65: 367. Micromachining
227 of quartz with ultrashort laser pulses. Varel, H.,
228 <https://doi.org/10.1007/s003390050593>
229
- 230 3 Ablation characteristics of Au, Ag, and Cu metals using a femtosecond Ti:sapphire
231 laser. Furusawa, K., Takahashi, K., Kumagai, H. et al. Appl Phys A (1999) 69(Suppl
232 1): S359. <https://doi.org/10.1007/s003390051417>
233
- 234 4 Ablation threshold dependence on pulse duration for copper. Hashida, D., Semerok A.
235 F., Gobert, O., Petite, G., Izawa, Y., Wagner, J. F. Appl. Sur. Sci (2002) 197, 862
236 [https://doi.org/10.1016/S0169-4332\(02\)00463-4](https://doi.org/10.1016/S0169-4332(02)00463-4)
237
- 238 5 Femtosecond time-resolved laser-induced breakdown spectroscopy for detection and
239 identification of bacteria: A comparison to the nanosecond regime. Baudelet, M.,
240 Guyon, L., Yu, J., Wolf J. P., Amodeo, T., Fréjafon, E., Laloï P. Journal of Applied
241 Physics 99, 084701 (2006); doi: 10.1063/1.2187107
242
- 243 6 R. E. Russo, X. Mao, J. J. Gonzalez, V. Zorba†, J. Yoo. Laser ablation in analytical
244 chemistry. Anal. Chem. (2013), 85, 13, 6162-6177,
245 <https://dx.doi.org/10.1021/ac4005327>
246
- 247 7 M. Guillermin, J. P. Colombier, S. Valette, E. Audouard, F. Garrelie, and R. Stoian.
248 Optical emission and nanoparticle generation in Al plasmas using ultrashort laser pulses
249 temporally optimized by real-time spectroscopic feedback. Phys. Rev. B (2010) 82,
250 035430
251
- 252 8 Forster, M., Égerházi, L., Haselberger, C. et al. Femtosecond laser interaction with
253 pulsed-laser deposited carbon thin films of nanoscale thickness Appl. Phys. A (2011)
254 102: 27. <https://doi.org/10.1007/s00339-010-6013-5>
255
- 256 9 L.M.Cabalín J.J.Laserna. Experimental determination of laser induced breakdown
257 thresholds of metals under nanosecond Q-switched laser operation. Spectrochimica Acta
258 Part B 53 (1998) 723-730 [https://doi.org/10.1016/S0584-8547\(98\)00107-4](https://doi.org/10.1016/S0584-8547(98)00107-4)
259
- 260 10 L.Torrìsi, S. Gammìno, L. Andò, V. Nassisi, D. Doria, A. Pedone. Comparison of
261 nanosecond laser ablation at 1064 and 308 nm wavelength. Applied Surface Science
262 (2003) 210(3):262-273. [https://doi.org/10.1016/S0169-4332\(02\)01467-8](https://doi.org/10.1016/S0169-4332(02)01467-8)
263
- 264 11 L. Torrìsi, A. Borrielli, D. Margarone. Study on the ablation threshold induced by
265 pulsed lasers at different wavelengths. Nucl. Instrum Meth. B (2007) 255(2):373-379
266 <https://doi.org/10.1016/j.nimb.2006.12.144>
267
- 268 12 A. Spiro, M. Lowe, G. Pasmanik. Drilling rate of five metals with picosecond laser
269 pulses at 355, 532, and 1064 nm. Appl. Phys. A (2012) 107: 801.
270 <https://doi.org/10.1007/s00339-012-6910-x>

271
272 13 A.A. Dergachev, A.A. Ionin, V.P. Kandidov, L.V. Seleznev, D.V. Sinitsyn,
273 E.S. Sunchugasheva, S.A. Shlenov. Filamentation of IR and UV femtosecond pulses
274 upon focusing in air. *Quantum Electron.* (2013) 43 29.
275 <https://doi.org/10.1070/QE2013v043n01ABEH015007>
276
277 14 J. Hoffman, J. Chrzanowska, S. Kucharski, T. Moscicki, I. N. Mihailescu, C.
278 Ristoscu, Z. Szymanski. The effect of laser wavelength on the ablation rate of carbon.
279 *Appl. Phys. A* (2014) 117, 395. <https://doi.org/10.1007/s00339-014-8506-0>
280
281 15 D. V. Apeksimov, O. A. Bukin, E. E. Bykova, Yu. E. Geints, S. S. Golik, A. A.
282 Zemlyanov, A. A. Il'in, A. M. Kabanov, G. G. Matvienko, V. K. Oshlakov, A. V.
283 Petrov, E. B. Sokolova. Filamentation of the focused Ti: Sapphire laser pulse in air at
284 two harmonics. *Plasma Phys. Rep.* (2013) 39: 1074.
285 <https://doi.org/10.1134/S1063780X13070106>
286
287 16 T. J. Wang, J. F. Daigle, J. Ju, S. Yuan, R. Li, S. L. Chin. Forward lasing action at
288 multiple wavelengths seeded by white light from a femtosecond laser filament in air.
289 *Phys. Rev. A* (2013) 88, 053429. <https://doi.org/10.1103/PhysRevA.88.053429>
290
291 17 Yu. E. Geints, A. A. Zemlyanov, A. A. Ionin, D. V. Mokrousova, L. V. Seleznev, D.
292 V. Sinitsyn, E. S. Sunchugasheva. Comparative analysis of post-focal filamentation of
293 focused UV and IR laser pulses in air. *Quantum Electron.* (2015) 45 321.
294 <https://doi.org/10.1070/QE2015v045n04ABEH015640>
295
296 18 N. L. LaHaye, S. S. Harilal, P. K. Diwakar, A. Hassanein, P. Kulkarni. The effect of
297 ultrafast laser wavelength on ablation properties and implications on sample
298 introduction in inductively coupled plasma mass spectrometry. *J. Appl. Phys.* (2013)
299 114, 023103. <http://dx.doi.org/10.1063/1.4812491>
300
301 19 S. M. Wiggins, J. Bonse, J. Solis, C. N. Afonso, K. Sokolowski-Tinten, V. V.
302 Temnov, P. Zhou, D. von der Linde. The influence of wavelength on phase
303 transformations induced by picosecond and femtosecond laser pulses in GeSb thin
304 films. *J. Appl. Phys.* (2005) 98, 113518. <https://doi.org/10.1063/1.2139830>
305
306 20 X. Zeng, X.L. Mao, R. Greif, R.E. Russo. Experimental investigation of ablation
307 efficiency and plasma expansion during femtosecond and nanosecond laser ablation of
308 silicon. *Appl. Phys. A* (2005) 80: 237. <https://doi.org/10.1007/s00339-004-2963-9>
309
310 21 A. E. Hussein, P. K. Diwakar, S. S. Harilal, A. Hassanein. The role of laser
311 wavelength on plasma generation and expansion of ablation plumes in air. *J. Appl.*
312 *Phys.* (2013) 113, 143305. <http://dx.doi.org/10.1063/1.4800925>
313
314 22 K. Sokolowski-Tinten, J. Bialkowski, A. Cavalleri, M. Boing, H. Schueler, D. von
315 der Linde. Dynamics of femtosecond-laser-induced ablation from solid surfaces. *Proc.*
316 *SPIE* 3343, High-Power Laser Ablation, (1998). <https://doi.org/10.1117/12.321593>
317
318 23 T. Y. Choi, C. P. Grigoropoulos. Plasma and ablation dynamics in ultrafast laser
319 processing of crystalline silicon. *J. Appl. Phys.* (2002) 92, 4918.
320 <https://doi.org/10.1063/1.1510565>

321
322 24 D. Puerto, W. Gawelda, J. Siegel, J. Bonse, G. Bachelier, J. Solis. Transient
323 reflectivity and transmission changes during plasma formation and ablation in fused
324 silica induced by femtosecond laser pulses. *Appl. Phys. A* (2008) 92, 803.
325 <https://doi.org/10.1007/s00339-008-4586-z>
326
327 25 S. Rapp, J. Rosenberger, M. Domke, G. Heise, H. P. Huber, M. Schmidt. Ultrafast
328 pump-probe microscopy reveals the mechanism of selective fs laser structuring of
329 transparent thin films for maskless micropatterning. *Appl. Surf. Sci.* (2014) 290, 30
330 368-372. <https://doi.org/10.1016/j.apsusc.2013.11.086>
331
332 26 M. Garcia-Lechuga, J. Siegel, J. Hernandez-Rueda, J. Solis. Femtosecond laser
333 ablation of dielectric materials in the optical breakdown regime: Expansion of a
334 transparent shell. *Appl. Phys. Lett.* (2014) 105, 112902.
335 <http://dx.doi.org/10.1063/1.4895926>
336
337 27 I. Carrasco-García, J. M. Vadillo, J. J. Laserna. Visualization of surface
338 transformations during laser ablation of solids by femtosecond pump-probe time-
339 resolved microscopy. *Spectrochimica Acta Part B* 113 (2015) 30–36.
340 <http://dx.doi.org/10.1016/j.sab.2015.08.009>
341
342 28 I. Carrasco-García, J. M. Vadillo, J. J. Laserna. Monitoring the dynamics of the
343 surface deformation prior to the onset of plasma emission during femtosecond laser
344 ablation of noble metals by time-resolved reflectivity microscopy. *Spectrochimica Acta*
345 *Part B* 131 (2017) 1–7. <http://dx.doi.org/10.1016/j.sab.2017.02.014>
346
347 29 B. Rethfeld, K. Sokolowski-Tinten, D. von der Linde, S. I. Anisimov. Timescales in
348 the response of materials to femtosecond laser excitation. *Appl. Phys. A* (2004) 79: 767.
349 <https://doi.org/10.1007/s00339-004-2805-9>
350
351 30 E.G. Gamaly, A.V. Rode. Physics of ultra-short laser interaction with matter: From
352 phonon excitation to ultimate transformations. *Progress in Quantum Electronics* 37
353 (2013) 215–323. <http://dx.doi.org/10.1016/j.pquantelec.2013.05.001>
354
355 31 E. G. Gamaly, A. V. Rode, B. Luther-Davies, V. T. Tikhonchuk. Ablation of solids
356 by femtosecond lasers: Ablation mechanism and ablation thresholds for metals and
357 dielectrics. *Physics of Plasmas* (2002) 9, 949. <http://dx.doi.org/10.1063/1.1447555>
358
359 32 A. Horn, I. Mingareev, J. Gottmann, A. Werth, U. Brenk. Dynamical detection of
360 optical phase changes during micro-welding of glass with ultra-short laser radiation.
361 *Meas. Sci. Technol.* 19 (2008) 015302. <https://doi.org/10.1088/0957-0233/19/1/015302>
362
363 33 P. Balling, J. Schou. Femtosecond-laser ablation dynamics of dielectrics: basics and
364 applications for thin films. *Rep. Prog. Phys.* 76 (2013) 036502.
365 <https://doi.org/10.1088/0034-4885/76/3/036502>
366
367 34 B. Rethfeld, A. Kaiser, M. Vicanek, G. Simon. Femtosecond laser-induced heating
368 of electron gas in aluminium. *Appl. Phys. A* (1999) 69 [Suppl.], S109–S112.
369 <https://doi.org/10.1007/s003399900228>
370

371 35 N.N. Nedialkov, P.A. Atanasov, S. Amoruso, R. Bruzzese, X. Wang. Laser ablation
372 of metals by femtosecond pulses: Theoretical and experimental study. Appl. Surf. Sci.
373 253 (2007) 7761–7766. <https://doi.org/10.1016/j.apsusc.2007.02.083>
374

375 36 E. G. Gamaly. The physics of ultra-short laser interaction with solids at non-
376 relativistic intensities. Physics Reports 508 (2011) 91–243.
377 <https://doi.org/10.1016/j.physrep.2011.07.002>
378

379 37 K. Sokolowski-Tinten, J. Bialkowski, A. Cavalleri, D. von der Linde. Observation of
380 a transient insulating phase of metals and semiconductors during short-pulse laser
381 ablation. Applied Surface Science 127–129 (1998). 755–760.
382 [https://doi.org/10.1016/S0169-4332\(97\)00736-8](https://doi.org/10.1016/S0169-4332(97)00736-8)
383

384 38 S. K. Sundaram, E. Mazur. Inducing and probing non-thermal transitions in
385 semiconductors using femtosecond laser pulses. Nature Materials (2002) 1, pages 217–
386 224.
387

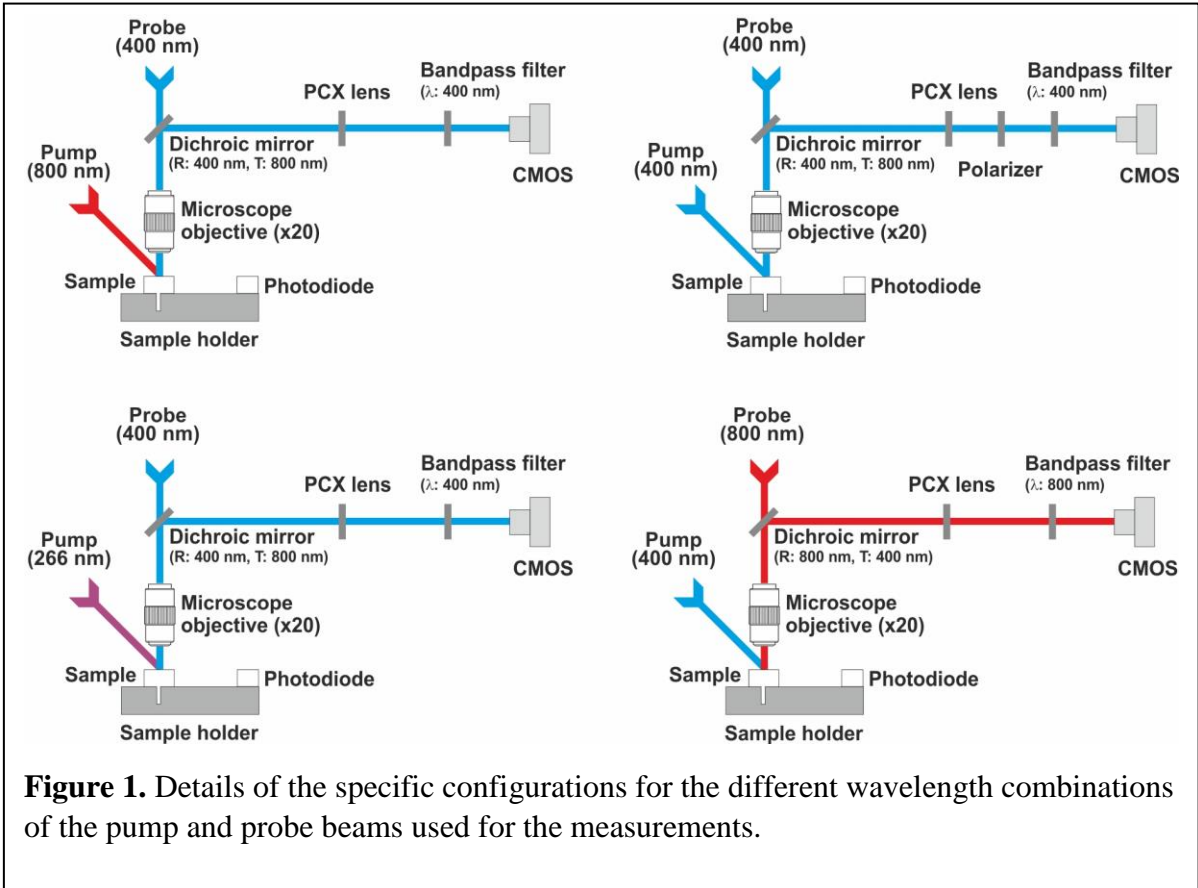
388 39 N. M. Bulgakova, I. M. Burakovi, Y. P. Meshcheryakov, R. Stoian, A. Rosenfeld, I.
389 V. Hertel. Theoretical Models and Qualitative Interpretations of Fs Laser Material
390 Processing. JLMN-Journal of Laser Micro/Nanoengineering (2007) 2, 1, 76-86.
391

392 40 M. V. Shugaev, C. Wu, O. Armbruster, A. Naghilou, N. Brouwer, D. S. Ivanov, T. J.
393 Y. Derrien, N. M. Bulgakova, W. Kautek, B. Rethfeld, L. V. Zhigilei. Fundamentals of
394 ultrafast laser–material interaction. Ultrafast Laser Synthesis and Processing of
395 Materials (2016) 41, 12, 960-968. <https://doi.org/10.1557/mrs.2016.274>
396

397 41 D. von der Linde, K. Sokolowski-Tinten. The physical mechanisms of short-pulse
398 laser ablation. Applied Surface Sci. 154–155 (2000). 1–10.
399 [https://doi.org/10.1016/S0169-4332\(99\)00440-7](https://doi.org/10.1016/S0169-4332(99)00440-7)
400

401 42 N. E. Bykovskii, Y. V. Senatskii. On the Formation Mechanism of Interference
402 Rings in the Ablation Area on the Condensed Medium Surface under Irradiation with
403 Femtosecond Laser Pulses. Phys. Solid State (2018) 60: 404.
404 <https://doi.org/10.1134/S1063783418020087>
405

406 43. Marina López-Claros, José M. Vadillo and J. Javier Laserna. Determination of plasma
407 ignition threshold fluence during femtosecond single-shot laser ablation on metallic
408 samples detected by optical emission spectroscopy. J. Anal. At. Spectrom., 2015, 30,
409 1730–1735. <https://doi.org/10.1039/c5ja00076a>
410
411



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418 **Figure 2**

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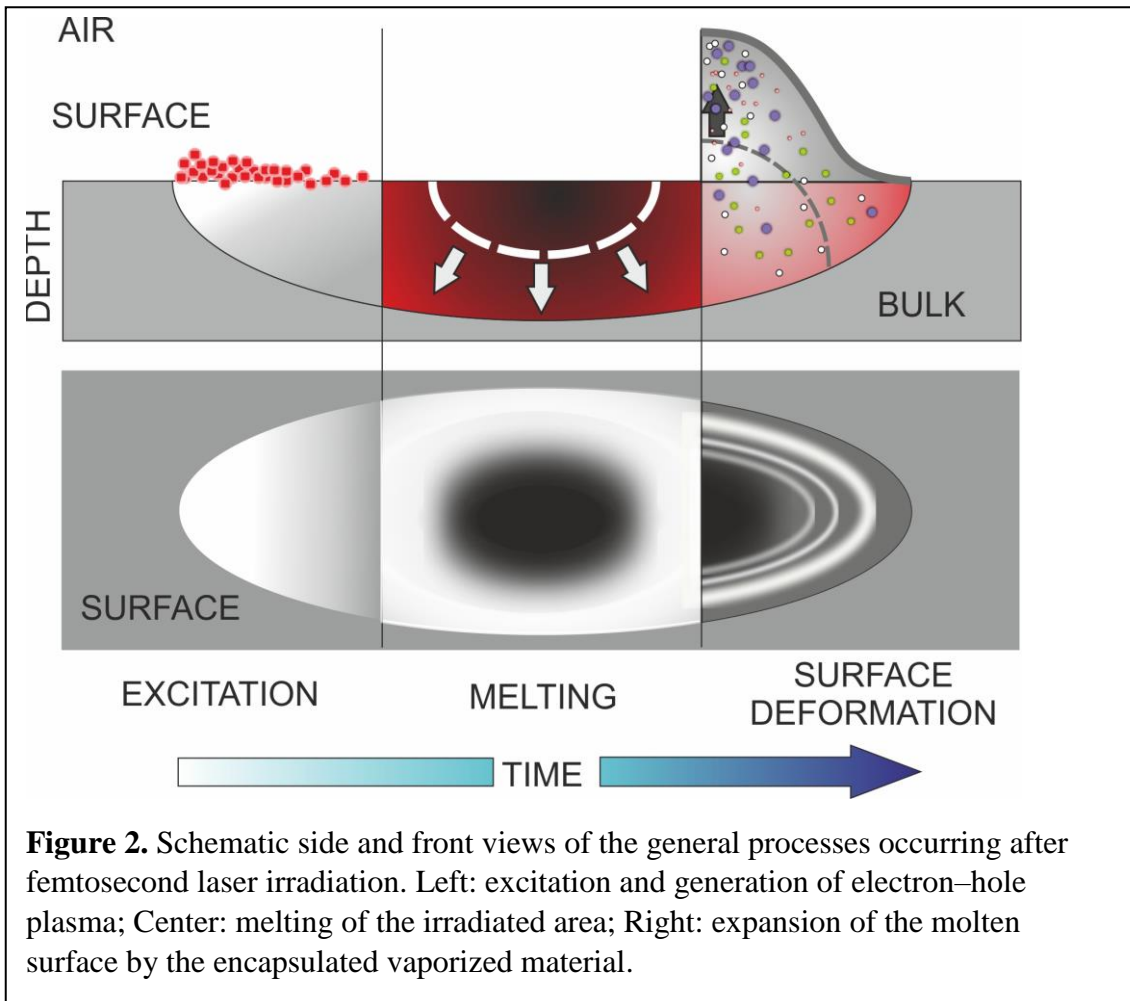
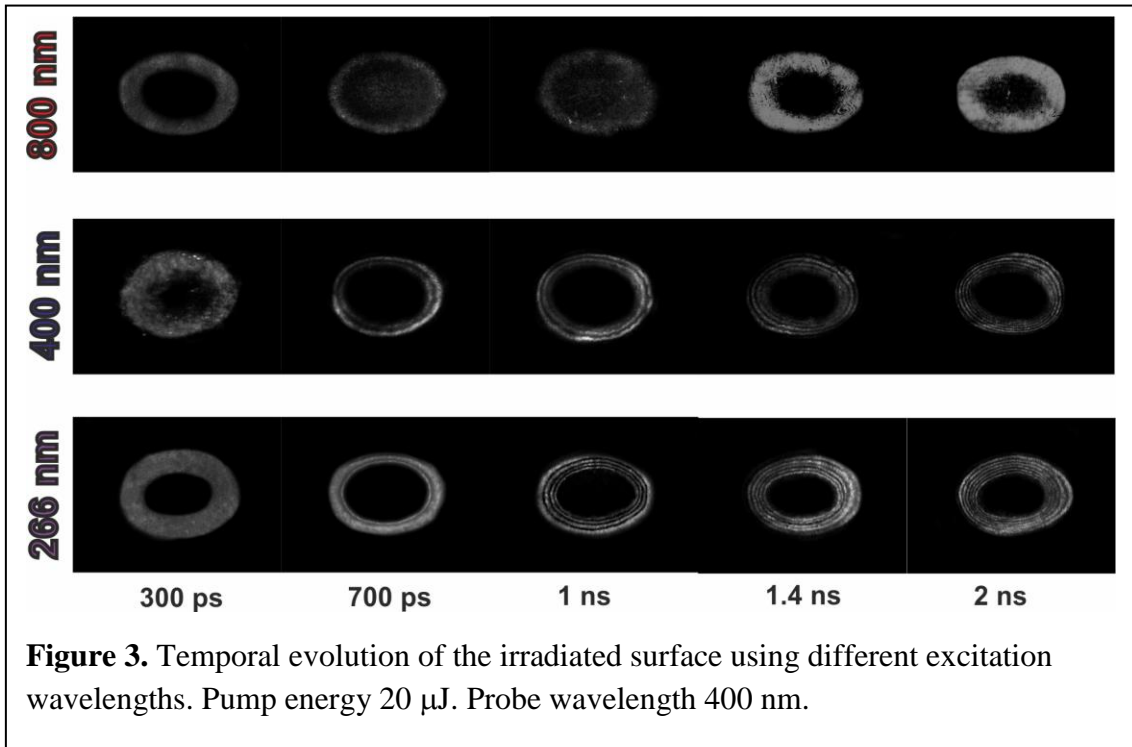


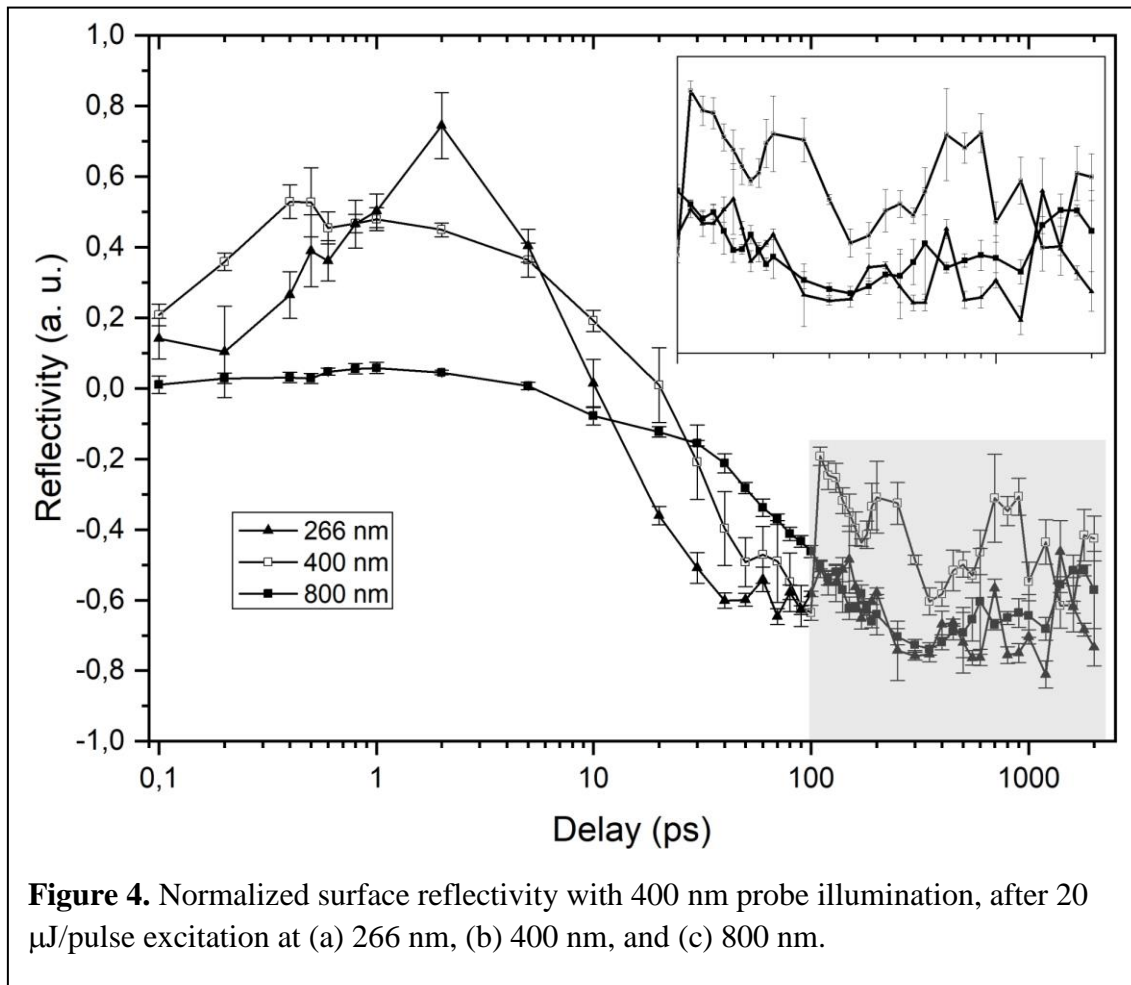
Figure 2. Schematic side and front views of the general processes occurring after femtosecond laser irradiation. Left: excitation and generation of electron–hole plasma; Center: melting of the irradiated area; Right: expansion of the molten surface by the encapsulated vaporized material.

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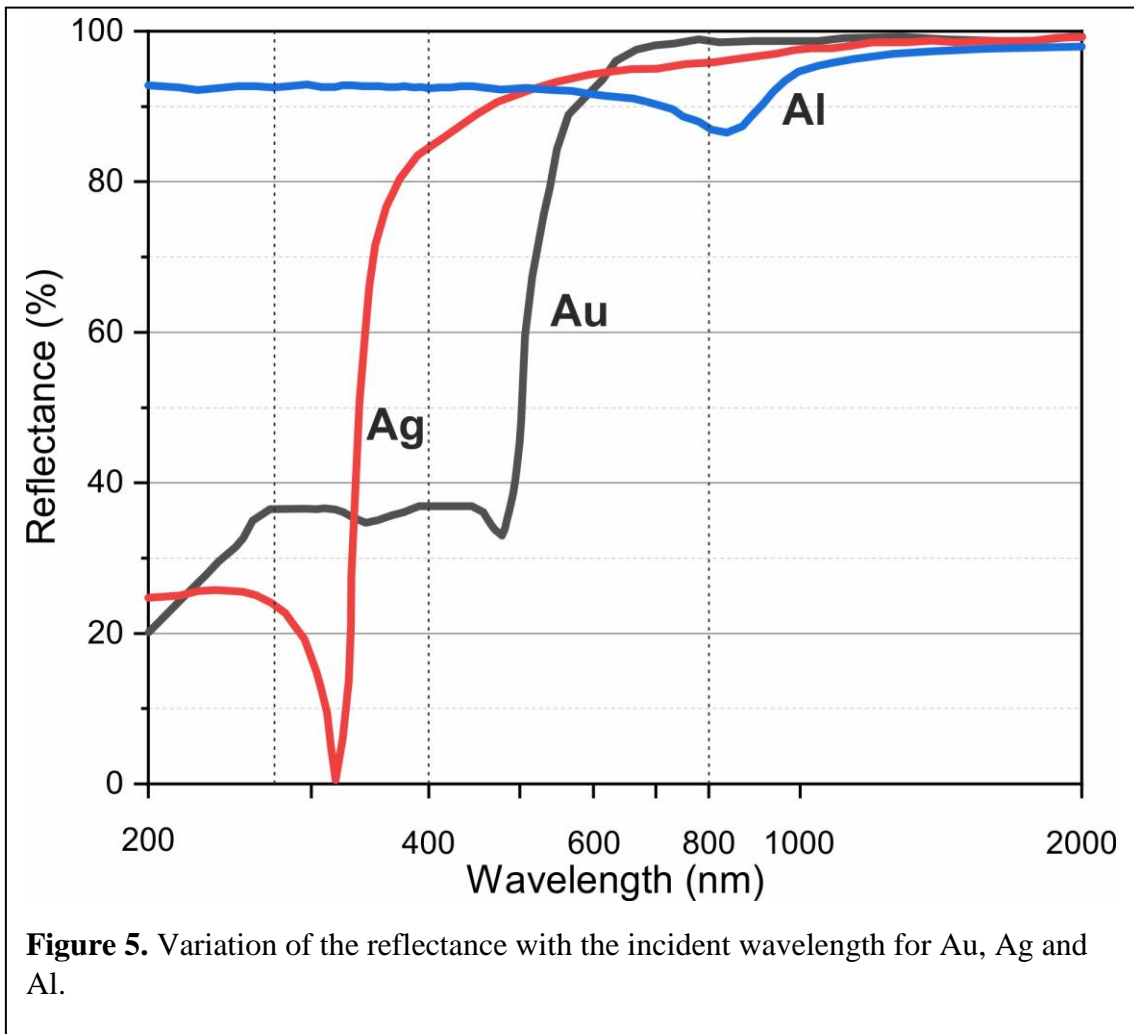
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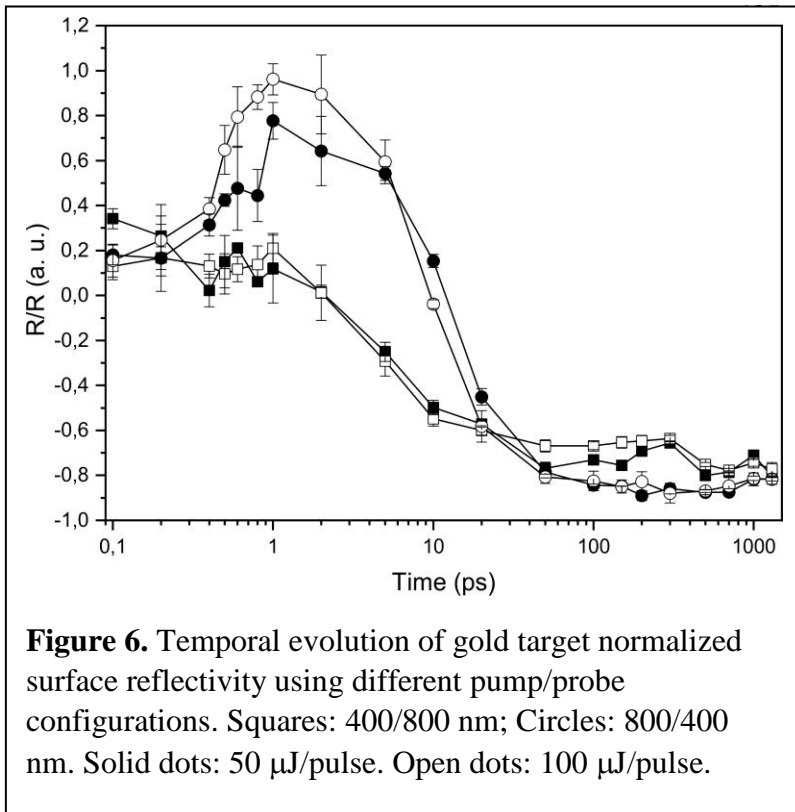
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453 **FIGURE CAPTIONS**

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455 **Figure 1:** Details of the specific configurations for the different wavelength
456 combinations of the pump and probe beams used for the measurements.

457

458 **Figure 2:** Schematic side and front views of the general processes occurring after
459 femtosecond laser irradiation. Left: excitation and generation of electron–hole plasma;
460 Center: melting of the irradiated area; Right: expansion of the molten surface by the
461 encapsulated vaporized material.

462

463 **Figure 3:** Temporal evolution of the irradiated surface using different excitation
464 wavelengths. Pump energy 20 μJ . Probe wavelength 400 nm.

465

466 **Figure 4:** Normalized surface reflectivity with 400 nm probe illumination, after 20
467 $\mu\text{J}/\text{pulse}$ excitation at (a) 266 nm, (b) 400 nm, and (c) 800 nm.

468

469 **Figure 5:** Variation of the reflectance with the incident wavelength for Au, Ag and Al.

470

471 **Figure 6:** Temporal evolution of gold target normalized surface reflectivity using
472 different pump/probe configurations. Squares: 400/800 nm; Circles: 800/400 nm. Solid
473 dots: 50 $\mu\text{J}/\text{pulse}$. Open dots: 100 $\mu\text{J}/\text{pulse}$.

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